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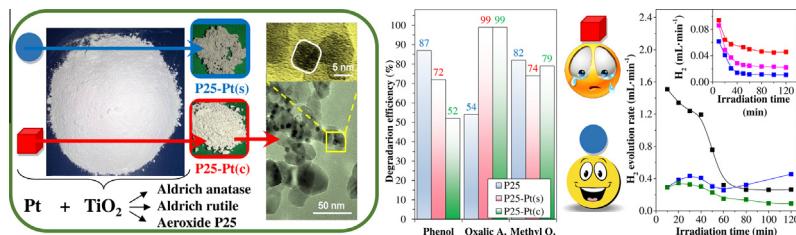
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Graphical abstract

Polyhedral Pt vs. spherical Pt nanoparticles on commercial titanias: Is shape tailoring a guarantee of achieving high activity?

pp xxx-xxx

G. Kovács, Sz. Fodor, A. Vulpoi, K. Schrantz, A. Dombi, K. Hernádi, V. Danciu, Zs. Pap ^{*}, L. Baia

Highlights

- The optical properties of P25 based composites are Pt-shape dependent.
- The phenol and methyl-orange degradation is shape and base catalyst dependent.
- Fine-tuning of the degradation intermediates is possible via Pt morphology.
- Kinetics of oxalic acid degradation was independent from the shape of Pt.
- The H₂ production was efficient in the case of spherical Pt with high index facets.



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3 **Q1** Polyhedral Pt vs. spherical Pt nanoparticles on commercial titanias: Is
4 **Q2** shape tailoring a guarantee of achieving high activity?

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15 **ARTICLE INFO** **ABSTRACT**

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photocatalytic activity/H₂ production results and the shape of the Pt nanoparticles.

Phenol and phenolic compounds are commonly used in various industries, such as agriculture and pharmaceutical and food industries. Among expensive and less efficient wastewater treatment methods, TiO₂-based heterogeneous photocatalysis can be used in a promising way to eliminate these kinds of organic compounds. An important aspect is the photodegradation of phenol itself (which was studied already in the early 1990s [36]), where various hydroxylated phenol compounds [37] can appear during the degradation process (such as pyrocatechol (PY), hydroquinone (HQ), and resorcinol (RES) [38]). These organic compounds, according to widely accepted safety protocols, are at the category 1 or 2 carcinogenic risk and toxicity levels [8,39]. Furthermore, these primary degradation products are more toxic than the phenol itself. Experiments performed on laboratory mice showed that hydroquinone is 2.3 times, while pyrocatechol is 1.4 times more toxic than the already mentioned phenol. The good news is that resorcinol is only 1.11 times less toxic. For the exact LD₅₀ values, Ref. [40] can be consulted.

The purpose of this study was to elucidate structural peculiarities via various investigation methods (diffuse reflectance spectroscopy (DRS), transmission electron microscopy (TEM), X-ray diffraction (XRD)) and to correlate them with the activity of the different types of commercial TiO₂ powders coupled with differently shaped Pt nanoparticles in terms of photodegradation, intermediates' evolution trends, and H₂ production. The research strategy is presented schematically in Fig. 1.

2. Experimental

2.1. Synthesis of the platinum nanoparticles

2.1.1. Materials

Ethylene glycol (EG, 99.8%, anhydrous), AgNO₃ (ACS reagent grade, >99.0%), H₂PtCl₆ (ACS reagent), polyvinyl pyrrolidone (PVP, Mw ≈ 40,000), ethanol (>99.8% reagent grade), acetone (>99.9%), trisodium citrate (ACS reagent grade, >99.0%), NaBH₄ (purum, >96%), and Aldrich anatase and Aldrich rutile reference photocatalysts were purchased from Sigma-Aldrich, while Aeroxide P25 was acquired from Evonik Industries and used without further purification.

2.1.2. Synthesis of the polyhedral Pt nanoparticles

The synthesis of the polyhedral Pt nanoparticles was based on a polyol method already available in the literature [41,42]. In a typical synthesis process, 8 mL EG and 1 mL AgNO₃ (0.04 M) were added in a three-neck flask and heated at 160 °C in a hot-oil bath. Meanwhile, two other solutions were prepared at room temperature: a 2 mL 0.025 M solution of H₂PtCl₆ (solution 1) and a 4 mL 0.375 M solution of PVP in EG (solution 2). These were added simultaneously to the reaction vessel as follows: 60 µL from solution 2 and 30 µL from solution 1 every 30 s. Afterward, the resultant mixture was refluxed at 160 °C for a further 25 min. After that, the product was centrifuged at 12,000 rpm for 15 min and washed four times with acetone and hexane. Finally, the obtained polyhedral Pt nanoparticles were redispersed in ethanol.

2.1.3. Synthesis of the spherical Pt nanoparticles

Into a specific reaction vessel 43 mL of ultrapure water was measured, followed by the addition of a 6.3 mL 5 mM solution of trisodium citrate. After 30 min, a 550 µL 22.8 mM H₂PtCl₆ solution was added and the mixture was stirred at room temperature for another 30 min. The last step in this synthesis was reduction by the addition of 1 mL 0.15 M NaBH₄. The reaction mixture was

stirred for 1 h to eliminate the by-products and the unreacted NaBH₄. The obtained platinum sol was then used immediately for the impregnation of the chosen titanias.

2.2. Synthesis of the TiO₂-Pt nanocomposites

The chosen commercial titania (Aldrich anatase – AA, Aldrich rutile – AR, and Evonik Aeroxide P25-P25) (400 mg) was suspended in 400 mL ultrapure water and sonicated for 15 min. Then the necessary quantity of Pt suspension was added to the homogenized dispersion under vigorous stirring. The added suspensions' volume was calculated in such a way that the Pt nanoparticles' weight fraction in the final composites' mass would be 1 wt.% (in all the Pt-containing composite materials – no significant Pt loss was detected during the preparation procedure). After 5 min of ultrasonically assisted homogenization and 20 min of vigorous stirring, the resulted suspension was dried at 80 °C for 24 h, resulting in a light gray/gray material. These powders were washed with ultrapure water (4400 rpm, 10 min) and dried again at 80 °C for 24 h.

The nomenclature of the samples was defined as follows: abbreviation of the base photocatalyst – Pt(s or c), where the first section can be defined as AA, AR, or P25, while in the second "s" stands for spherical and "c" for cuboctahedral (the dominant shape among the polyhedral Pt nanoparticles).

2.3. Methods and instrumentation

2.3.1. Characterization methods

X-ray diffraction (XRD) measurements were performed on a Shimadzu 6000 diffractometer using Cu K α radiation ($\lambda = 1.5406 \text{ \AA}$) equipped with a graphite monochromator. The anatase-rutile phase ratio in TiO₂ was evaluated by method used by Banfield [43], and the crystallites' average size was calculated using the Scherrer equation [44].

A JASCO-V650 spectrophotometer with an integration sphere (ILV-724) was used for measuring the DRS (diffuse reflectance spectroscopy) spectra of the samples ($\lambda = 300$ –800 nm). The possible electron transitions were evaluated by plotting $dR/d\lambda$ vs. λ , where R is the reflectance and λ is the wavelength [8,39,45], while the indirect band-gap of the photocatalysts was determined via the Kubelka-Munk method.

TEM/HRTEM images were obtained with a FEI Tecnai F20 field emission high-resolution transmission electron microscope operating at an accelerating voltage of 200 kV and equipped with an Eagle 4k CCD camera.

2.3.2. Assessment of the photocatalytic efficiencies

A photoreactor system with $6 \times 6 \text{ W}$ fluorescent lamps ($\lambda_{\text{max}} \approx 365 \text{ nm}$, irradiation time = 2 h) was used to measure the photocatalytic activities. The photocatalyst suspension containing the pollutant (initial concentration of phenol $c_0, \text{phenol} = 0.5 \text{ mM}$ or oxalic acid $c_0, \text{oxalic acid} = 5 \text{ mM}$ or methyl orange (MO) $c_0, \text{MO} = 125 \mu\text{M}$; catalyst concentration $c_{\text{photocatalyst}} = 1.0 \text{ g L}^{-1}$; total volume of the suspension $V_{\text{susp}} = 100 \text{ mL}$) was continuously purged with air to keep the dissolved oxygen concentration constant during the whole experiment. The concentration decrease of the chosen organic substrate (phenol and oxalic acid) and the phenol's primary degradation intermediates were followed using an Agilent 1100 series HPLC system (instrumental details can be found in Refs. [39,46], while details regarding the intermediate detection are detailed in Supporting information, Fig. S1). The concentration of MO was followed using a JASCO V-650 spectrophotometer at 513 nm. The assessed error of the photocatalytic tests (based on reproducibility experiments) was 2–5%, while in all cases

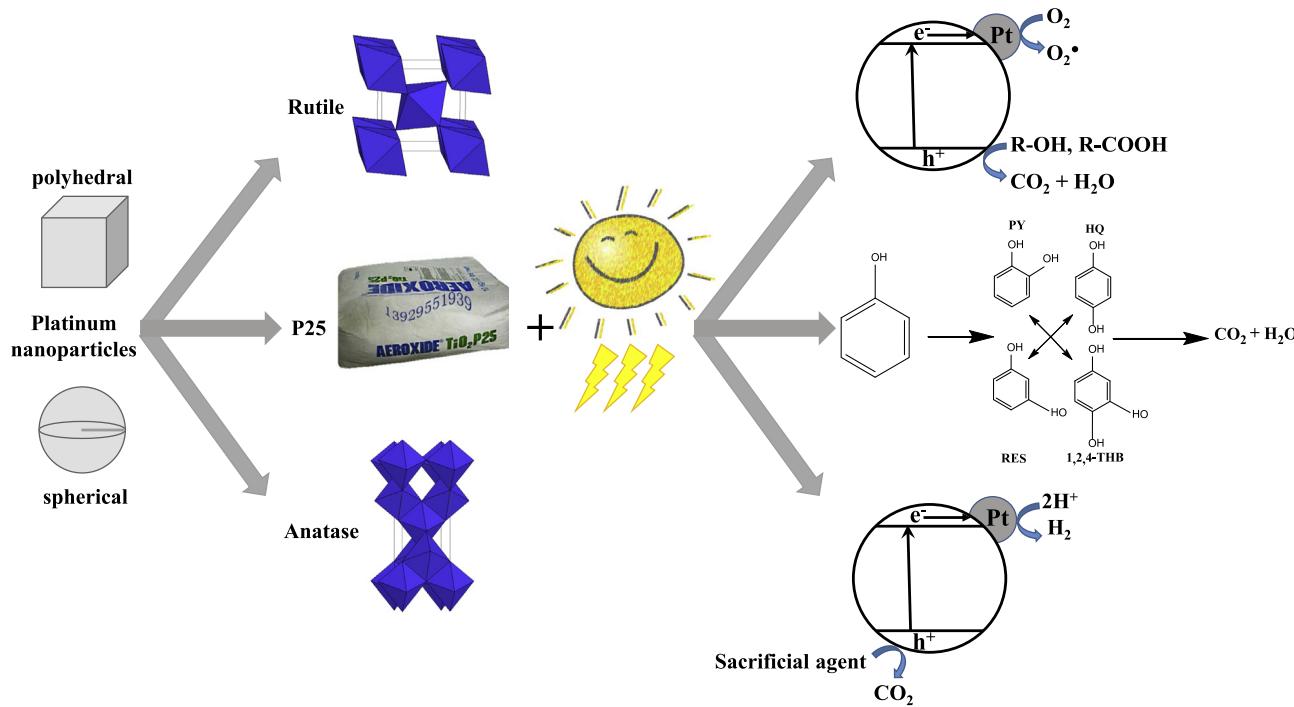


Fig. 1. Schematic diagram of the research methodology applied in the current investigations of the TiO_2 -Pt nanocomposites.

206 the adsorption phenomenon (at the used concentration values)
207 was negligible (below 5%).

208 To quantify the intermediates' evolution efficiently, the inter-
209 mediate evolution index (IEI) was introduced, which was calcu-
210 lated, using the following formula, where F_{int} is the empirical
211 intermediate concentration evolution function and C_{phendef} is the
212 quantity of phenol degraded in the case of the less well performing
213 catalyst. Therefore, a low IEI value means that the degradation
214 intermediate does not accumulate in the reaction system [8,39]:

$$217 \text{IEI} = \int_0^{C_{\text{p_hendef}}} F_{\text{int}} dC$$

218 2.3.3. Photocatalytic hydrogen production

219 The hydrogen production experiments were executed in a Pyrex
220 glass photoreactor thermostated at 25 °C and surrounded by ten
221 15 W low-pressure mercury lamps ($\lambda_{\text{max}} \approx 365$ nm). The suspen-
222 sion's concentration was 1.0 g L⁻¹ and the applied sacrificial agent
223 was oxalic acid (50 mM). During the photocatalytic runs the suspen-
224 sion was continuously purged with N₂ (50 mL min⁻¹) to avoid
225 the presence of O₂. The H₂ gas evolved was determined with a
226 Hewlett-Packard 5890 gas chromatograph equipped with a thermal
227 conductivity detector. On the basis of the H₂ concentrations
228 determined by GC from the flow rate of the N₂, the rate of H₂ evo-
229 lution (r) at the time of the sampling has been determined. The
230 total amount of hydrogen produced was estimated by integrating
231 the area under the hydrogen evolution curve using Origin 9 soft-
232 ware. The duration of the experiment was 2 h.

233 3. Results and discussion

234 3.1. Commercial titanias used – the research strategy

235 In the present work three well-known commercial titanias were
236 chosen: Aldrich anatase (AA), Aldrich rutile (AR), and Evonik
237 Aeroxide P25 (P25). They have been studied in detail during the
238 past 20–30 years and nearly all their major properties have been

239 uncovered (including surface quality, crystallinity-related issues,
240 and synergism of anatase and rutile phases [47]). Consequently,
241 they are ideal supports (without unknown parameters) for
242 investigating the effect of different-shaped platinum nanoparticles.
243 The research possibilities further exploited in the present work are
244 as follows:

- 245 • Anatase vs. rutile-both AA and AR are made from pure anatase
246 and rutile, while their average crystallite size is in the same
247 range
 - 248 • differentiate the effect of the titania crystal phase when
249 depositing specifically shaped Pt nanoparticles;
 - 250 • emphasize the importance of the Pt shape if it is deposited at
251 the surface of the same crystal phase (either on AA or on AR);
 - 252 • native mixture of crystal phases (meaning that the two crys-
253 tal phases are obtained during the same synthesis process) –
254 P25.
- 255 • Small vs. large crystallites – while P25 shows an average crys-
256 tallite size of 25–30 nm, both AA and AR contain nanocrystals
257 between 150 and 300 nm
 - 258 • electron-transfer-related issues between Pt and titania
259 nanocrystals;
 - 260 • electron-transfer-related issues if different-shaped Pt
261 nanocrystals are deposited.

265 3.2. TiO_2 -Pt composites: why different-shaped platinum 266 nanoparticles?

267 Based on the above-mentioned strategy, polyhedral (domi-
268 nantly cuboctahedral) and spherical particle geometries were cho-
269 sen to illuminate the importance of the shape of platinum
270 nanocrystals for the photocatalytic activity of TiO_2 -Pt
271 nanocomposites.

272 In other research fields such as electrocatalysis, there is already
273 significant work regarding the influence of the crystal shape of the
274 noble metals. Tian et al. have [48] already shown that in electro-
275 oxidation processes the shape of the Pt nanoparticles is crucial,

due to the number of the so-called stepped atoms, which can be found at the high-indexed crystal facets ((730), (411), etc.) and are responsible for the enhanced electrocatalytic activity [49]. Thus, the motivation of the present work is to illuminate the same aspects for the photocatalytic processes.

3.3. *TiO₂-Pt nanocomposites: characterization*

The base photocatalysts' crystal phase composition, crystal size, and specific surface area values are summarized in Table 1. The parameters obtained from the measurements coincide with the ones given by the manufacturer or with those published in the literature [50,51] (Fig. 2). By depositing platinum on the surface of these materials, no structural changes were observed, as expected. The next step in the characterization of these materials was to literally study the morphology of these nanocomposites.

First the morphology of the polyhedral platinum nanoparticles was examined by HRTEM while the lattice fringes were evaluated based on Refs. [41,52]. The obtained micrographs are presented in Fig. 3. As expected, the dominant shape of the nanocrystallites was cuboctahedral/octahedral – 72% (a relatively small percentage of tetrahedral – 5% – and some undefined polyhedral particles – 23% – were also noticed).¹ The interplanar distances were evaluated by FFT. The size distribution of these platinum nanocrystallites (both spherical and polyhedral ones) was homogeneous, most of them having a size of 4–6 nm (85%), as illustrated in Fig. 3.

The deposition of the platinum nanoparticles at the surfaces of the commercial titanias was also successful, as shown by Fig. 3. While in the case of P25 it was quite easy to obtain high-quality images of the deposition of platinum nanoparticles, the situation was dire in the case of AA and AR due to their large crystal size (200–300 nm). This is why only P25-related TEM micrographs were presented.

3.4. *The TiO₂-Pt nanocomposites: optical properties*

One of the first aspects that need investigation for materials with photocatalytic potential is their optical properties. The first, simplest approach was to examine the obtained nanocomposites' color. One may expect that the color of the composite materials should not change at all when the nanocrystals' shape is varied, because in each case we have the same material (the same optical "property set" should be observable) with the same composition.

However, as can be clearly seen in Fig. 4, just by changing the shape of the platinum nanoparticles, while using the same base catalyst (P25), an interesting change occurred in the investigated nanocomposites' color (intense creamy gray for sample P25-Pt(c), conventional gray for P25-Pt(s)). These observations indicate that a more detailed study of the optical properties of these materials was inevitable.

To get quantified information about the optical peculiarities of these materials, the DRS and the first-order derivative DRS spectra were recorded (Fig. 5a) and the band-gap values calculated (Table 1). The AA-based composites were examined in the first step, to gain critical information when only a single crystalline phase of titania was present in the composite. As Pt nanoparticles are deposited onto the surface of AA, the band-gap value remains constant. This can be even more precisely observed in the first derivative spectra; the peak located at 375 nm (3.3 eV) in the case of AA does not shift at all in the platinum-containing composites (AA-Pt(s), AA-Pt(c)). This means that the possible electron transitions between the valence band and the conduction band are

¹ The shape distribution was estimated based on 10 TEM images – 150 particles acquired from 10 randomly selected spots on the used copper grid.

taking place within an electronic band system in which the band gap has the same value, without being influenced by the presence of platinum. Although their band gaps may not differ, the color change of the material is obvious from the rest of the DRS spectra. The situation slightly changed in the case of AR-based samples. By depositing Pt onto the surface of AR, the band-gap energy values are slightly changed from 2.96 to 2.91 and 2.82 eV (AR-Pt(s) and AR-Pt(c) composites). The mentioned changes are also faintly visible in the first derivative spectra.

The investigations in the cases of AA- and AR-based composites already suggest that in the case of P25 (where both anatase and rutile are present in a well-defined ratio) a mixture of the effects should be observable. As expected, the presence of Pt modified the optical properties of P25 significantly. The bare catalyst exhibits two electron transition bands in the first derivative DRS spectra, one assigned to anatase and the other to rutile (Fig. 5b). As polyhedral nanoparticles are deposited (P25-Pt(c)) at the surface of the material, the ratio of the two bands changes in favor of anatase, while the peak positions do not vary. If the deposited Pt nanoparticles are spherical (P25-Pt(s)), then the ratio of the anatase/rutile bands is even more balanced toward the anatase phase. This means that in the case of P25-based composites the presence of Pt denies/inhibits electron transitions within the rutile particles. If this is true, then an activity decrease should be observable for nonadsorbing pollutant degradation, such as phenol.

3.5. *The photocatalytic activity of the obtained nanocomposites*

3.5.1. *The photodegradation of phenol*

Some hints regarding the possible importance of the Pt crystal geometry are already given by the interesting changes observed in the optical properties of the composite materials (see Fig. 6).²

In the first instance the P25-based composites' activity was evaluated. It is known that this commercial powder is a versatile and quite efficient photocatalytic material, which can be seen also in the present case by achieving 87% of phenol decomposition in 2 h. As platinum nanoparticles were deposited on P25, the activity decreased significantly (achieving 72 and 52% of degraded phenol for samples P25-Pt(s) and P25-Pt(c)). This activity drop in the case of P25-based composites could have several causes. One could be the efficiency of the electron transfer processes. One hint regarding this was already given by the optical properties of the P25-based composites. It was shown that when Pt nanoparticles were deposited, the electron transition band (in the first derivative DRS spectra) corresponding to the rutile phase diminishes significantly, suggesting that a fraction of the electron transitions are "lost"/not happening at all. There is also a significant difference in phenol degradation yield (72% for P25-Pt(s) vs. 52% for P25-Pt(c)) between the two Pt-containing composites, and a further change can be noticed in the ratio of the anatase and rutile electron transition bands in favor of anatase in the case of composite P25-Pt(c). The latter phenomenon raises the possibility of a special interaction between rutile and Pt nanopolyhedra, which may be clarified in the section regarding AR based composites.

Pure AA itself proved to be quite active in the degradation of phenol, although the manifested degradation yield is inferior to that of P25 (63% vs. 87%). Based on the behavior of P25, it was expected that after platinum deposition the activity would further decrease, but surprisingly this was not the case. Both spherical and polyhedral Pt nanoparticles enhanced with a factor of 1.5 the

² Please note that the photocatalytic performance will be discussed based on the photocatalytic efficiency given in the percentage of phenol removed. This was chosen because in some of the cases the kinetics of the degradation changes abruptly; thus a clear evaluation of the activity based on reaction rates would be uninformative (just for comparison, the values are given in Table 1).

Table 1Main structural properties and photocatalytic performance of the obtained TiO_2 -Pt nanocomposites.

Sample	Crystal phase composition (wt.%)			Band gap (eV)	Degradation rate/yield (mmol \times min $^{-1}$ \times dm $^{-3}$)/ (%)			H_2 (mL) ^a	IEI values ($\times 10^{-6}$) for phenol			
	Anatase	Rutile	Pt		Phenol	Oxalic acid	Methyl orange		HQ	PY	RES	THB
AA	100/>150	–	–	3.26	$3.24 \times 10^{-3}/63$	–	$1.15 \times 10^{-3}/76$	0.0	5180	6300	–	80
AA-Pt(c)	99/>150	–	1/5	3.20	$7.17 \times 10^{-3}/91$	–	$1.03 \times 10^{-3}/78$	2.2	4590	5653	–	126
AA-Pt(s)	99/>150	–	1/5	3.18	$7.89 \times 10^{-3}/99$	–	$0.71 \times 10^{-3}/34$	40.7	295	530	590	–
AR	t. a.	99/≥150	–	2.96	$5.59 \times 10^{-3}/41$	–	$0.58 \times 10^{-3}/50$	0.0	5300	5465	–	104
AR-Pt(c)	t. a.	98/≥150	1/5	2.91	$6.26 \times 10^{-3}/46$	–	$0.23 \times 10^{-3}/25$	3.8	2726	2580	92	37
AR-Pt(s)	t. a.	98/≥150	1/5	2.82	$8.54 \times 10^{-3}/83$	–	$1.36 \times 10^{-3}/68$	22.8	600	200	104	–
P25	89/25	11	–	3.11	$9.28 \times 10^{-3}/87$	$28.8 \times 10^{-3}/54$	$1.25 \times 10^{-3}/82$	0.0	3247	1420	215	65
P25-Pt(c)	88.5/25	10.5/40	1/5	2.95	$3.14 \times 10^{-3}/52$	$117.4 \times 10^{-3}/100$	$1.33 \times 10^{-3}/74$	6.2	2780	1263	114	407
P25-Pt(s)	88.5/25	10.5/40	1/5	2.66	$5.75 \times 10^{-3}/72$	$124.4 \times 10^{-3}/100$	$2.17 \times 10^{-3}/79$	78.3	8230	4051	–	550

^a The total amount of hydrogen produced during the 2 h irradiation (calculated at standard conditions – 25 °C and atmospheric pressure).

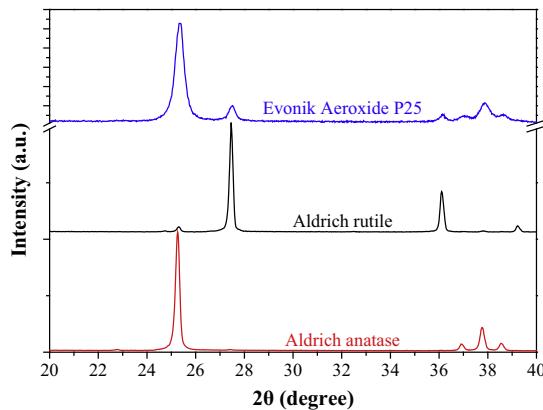


Fig. 2. XRD patterns of the three commercial titanias used as base photocatalysts throughout the current research.

activity of AA, attaining an amazing 99% (AA-Pt(s)) and 91% (AA-Pt(c)) phenol decomposition efficiency, which was equal/superior to the performance of bare P25. The optical reason for the observed activity enhancement can be totally ruled out, because the band gap of the AA-based materials does not change at all and the first derivative DRS spectra do not show any shifts in the electron transition energy ranges. The reason for the observed phenomenon could be the number of contacts between the composite components. Because AA crystals are relatively large (>150 nm) compared to P25 (25–40 nm), the effective number of Pt nanoparticles that could realize physical contact with an AA particle is very large. Consequently, the charge carriers generated could live longer because of the more efficient charge separation process (photogenerated electrons → more Pt nanoparticles). A similar enhancement mechanism was also proposed in our recent publication concerning AA/carbon nanotube composites [53].

The AR itself is a poor photocatalyst, showing only 41% phenol decomposition efficiency. By the deposition of polyhedral Pt nanoparticles (sample AR-Pt(c)), the situation remains nearly unchanged (41% and 46% removal efficiency). However, when the Pt nanocrystals were spherical, the activity jumped to 83% degradation efficiency. The surprising results obtained in the case of AR uncover important aspects of the functioning of these materials. Namely, it was already known that depositing spherical Pt on the surface of rutile enhances the photocatalytic activity [54], by the same principle as for AA (discussed previously) or for the reasons invoked by other authors, such as efficient light utilization above 400 nm [54]. However, polyhedral Pt nanoparticles do not show any effect on the activity of AR. This could be possible only if a charge transfer barrier existed between the two types of particles.

More precisely, Pt cuboctahedral possess (100) crystallographic planes, which are the least effective facets in electron transfer processes, while spherical Pt particles possess also a large number of high-index crystal facets, along with (100) [49]. This observation also supports the fact that in the case of P25-Pt(c) composite the rutile electron transition band's ratio shrinks considerably.

3.5.1.1. Degradation intermediates. The degradation intermediates of a specific organic pollutant can be a quite important factor when a photocatalyst reaches the doorstep of applicability. As already discussed in our recent papers, the fine tuning of the structure of a photocatalyst can lead to a major change in the ratio of the different degradation intermediates [8,39]. This also could be true if the shape of the platinum nanoparticles were changed in TiO_2 -Pt composites. Unfortunately, the less toxic primary degradation intermediate, resorcinol, was scarcely present during the degradation series, as phenol is attacked by the OH radical in *ortho* and *para* positions (Table 1). That is why 1,3,4-trihydroxybenzene was also present in a relatively small amount (Table 1). The following paragraphs will share details regarding HQ and PY (see Fig. 7).

The first observation that can be made is of the general influence of the Pt nanoparticles' presence. As these nanoparticles appeared on the surfaces of the commercial semiconductors, the registered IEI number decreased significantly (in some cases even an 18-fold decrease was observed for AA vs. AA-Pt(s); see Table 1). This means that the presence of Pt is beneficial from this point of view, because the toxic intermediates cannot accumulate [40]. The beneficial effect of platinum was valid only in the case of AA- and AR-based composite materials. In the case of P25 the IEI numbers increased significantly (~2.5-fold increase in the case of HQ and PY, composite P25-Pt(s); see Table 1), with a significant concomitant activity decrease (Table 1).

There are also important differences between the different-shaped Pt-containing nanocomposites. In the case of AA and AR, the presence of spherical Pt nanoparticles was more beneficial, considering the IEI number. However, the situation changed in the case of P25, where the recorded IEI number registered was much higher for P25-Pt(s) than for P25-Pt(c) (Table 1).

The results listed suggest two different conclusions. The first is referring to the pure crystalline phases of semiconductors, such as AA and AR; the presence of Pt diminishes the IEI number and in any case sphere-shaped Pt nanoparticles are the most efficient in this respect. The second was that, when smaller semiconductor nanoparticles are used and they are a mixture of two crystal phases (Evonik Aeroxide P25), the situation turns around. Consequently, the facts listed here opened up numerous research possibilities including the investigation of the Pt shape-crystal phase composition-crystal size relation triangle.

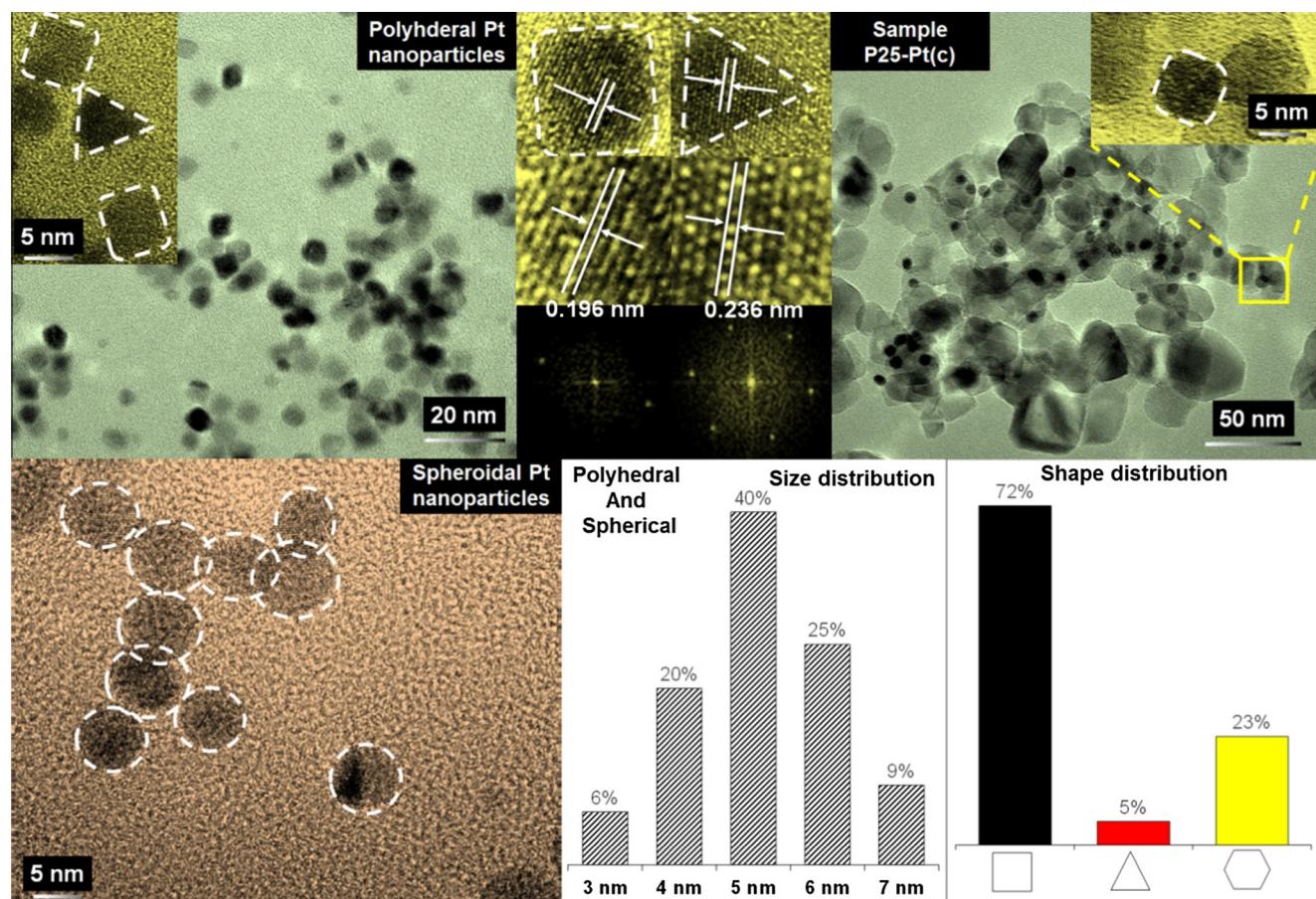


Fig. 3. TEM/HRTEM micrographs of the spherical and polyhedral Pt nanoparticles. TEM/HRTEM images of sample P25-Pt(c). The bar graphs show the size and shape distributions of the individual platinum nanoparticles. The zone axes were [001] and [111] (octahedral and tetrahedral particles).

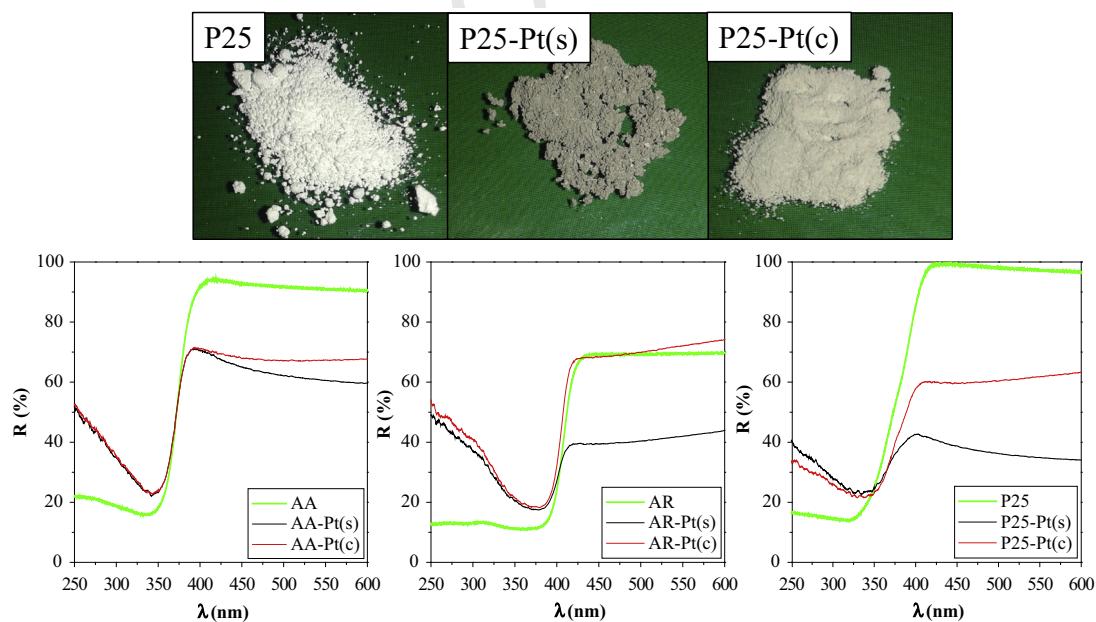


Fig. 4. Photographs of samples P25, P25-Pt(s) and P25-Pt(c) and the DRS spectra of the studied composite materials.

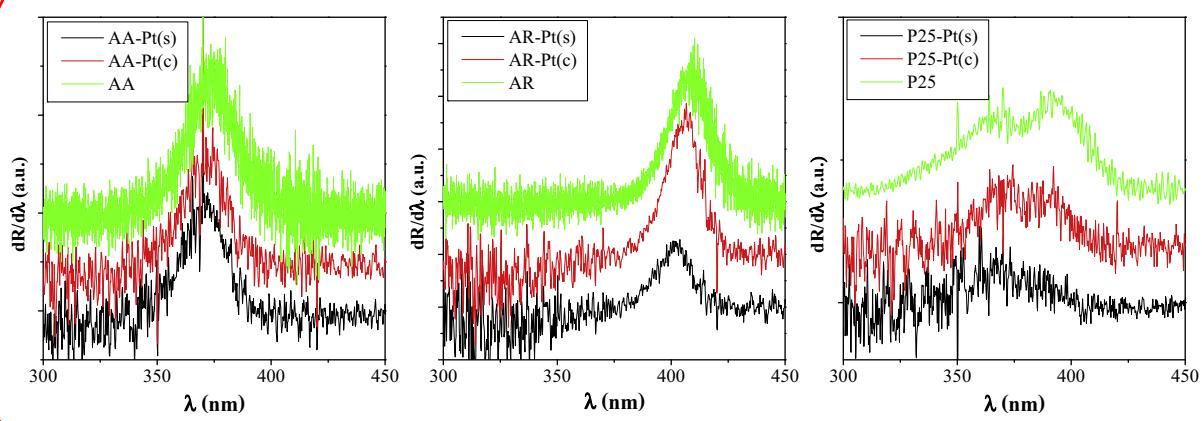
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3.5.2. The photodegradation of methyl orange (MO)

For the evaluation of the results obtained from the MO degradation, the research methodology used in the case of phenol was also

applied. The main research target was to observe the shape and base catalyst dependence of the degradation efficiencies. In the literature the degradation of MO is very well known, also with

(a) The first derivative DRS spectra of the composites made from the combination of P25, AA and AR with spherical Pt and polyhedral Pt nanoparticles.



(b) The deconvolution of the P25 based composites first derivative DRS spectra

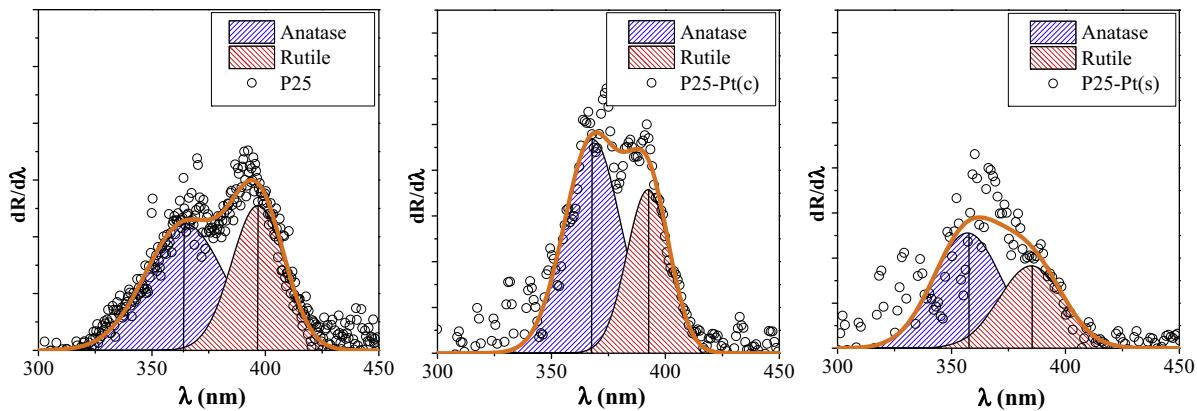


Fig. 5. (a) The first derivative DRS spectra of the composites made from the combination of P25, AA, and AR with spherical and polyhedral Pt nanoparticles. (b) The deconvolution of the P25-based composites' first derivative DRS spectra.

semiconductor oxide/Pt nanocomposites [55–57]. In these cases the papers point out the superior efficiency of platinum-containing nanocomposites.

When P25 is the base photocatalyst, there are quite small differences in the values of the MO conversion (between 75% and 80%). The most efficient was bare P25 (similarly to the degradation of phenol), followed by P25-Pt(c) and P25-Pt(s) in the means of conversion, although considering the initial reaction rates the most effective composite was P25-Pt(s).

In the case of AA-based composite materials, the sample AA-Pt(s) was the least effective (Table 1), while AA-Pt(c) achieved nearly the same reaction rate and degradation yield as the base photocatalyst AA (Table 1). These results are opposed to the conclusions drawn in the case of the phenol degradation experiments. This suggests that in the present case the large number of Pt-TiO₂ contacts are not sufficient to promote/enhance the photocatalytic activity. Consequently, besides the shape and the contact number as important factors, other parameters should be considered (surface complexation, mediated photodegradation, dynamic competition with intermediates for the photocatalyst's surface), which are currently under investigation and do not constitute the subject of the present paper.

In the case of AR-based composite photocatalysts, the analogy with phenol degradation is clearly visible. The AR-Pt(c) composite was less efficient than the base photocatalyst, while the composite containing spherical Pt nanoparticles (AR-Pt(s)) was more efficient than AR (Fig. 8 and Table 1).

As can be seen also in the case of MO, the shape of the Pt nanoparticles played an important role in defining the composite materials' activity. This observation emphasizes that the case of phenol degradation, where the noble metal shape can tailor the activity of a photocatalyst toward a given substrate, is not singular or exceptional.

3.5.3. Substrate dependence of the shape influence using a single type of base catalyst (P25)

As already discussed, it can be seen that both the shape of the platinum nanoparticles and the nature of the base catalyst are critical in every respect. Two different substrates have been investigated (MO and phenol), but both of them are poor at adsorbing pollutants. However, to get a complete picture regarding the activity spectrum of the shape-tailored composites, a well-known well absorbing organic substrate should be chosen, such as oxalic acid, which we have used successfully in other recent work [46,50]. In the present case P25 was chosen because it is the commercial photocatalyst that is used most frequently in photocatalysis-related publications [47].

In Fig. 9 it can be seen that degrading phenol with Pt-modified P25 leads to inhibition of the photocatalytic activity from 87% to 72% of degraded phenol (as discussed in Section 3.5). Also, by using polyhedral (cuboctahedral) nanoparticles, the activity was decreased further to 52% of degraded phenol, as discussed in the appropriate section of the paper. Interestingly, in the case of MO (despite the fact that it is a poorly adsorbing substrate), the Pt

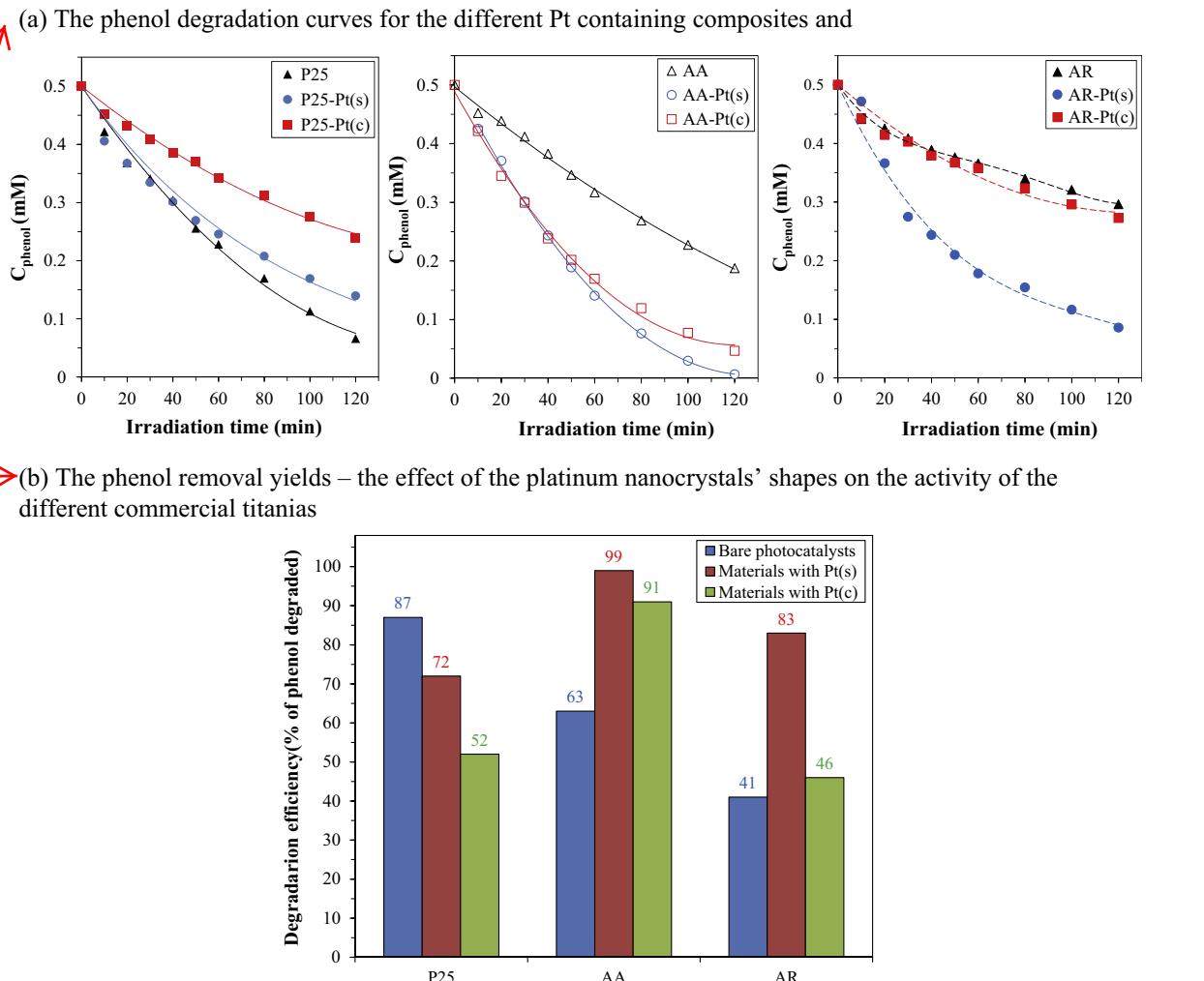


Fig. 6. (a) The phenol degradation curves for the different Pt-containing composites. (b) The phenol removal yields: Effects of the platinum nanocrystals' shapes on the activity of the different commercial titanias.

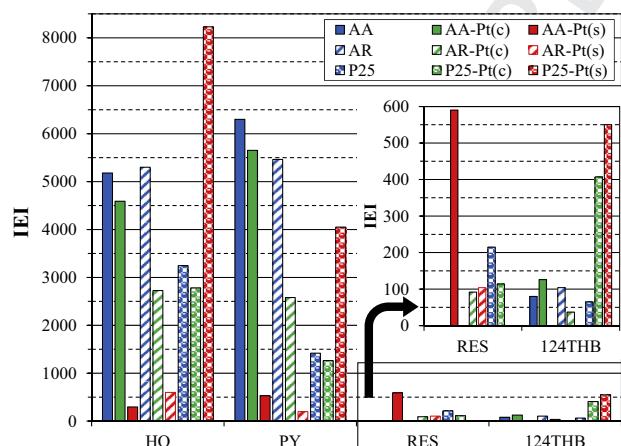


Fig. 7. The degradation intermediates' presence evaluated in terms of the IEI value.

nanoparticles did not inhibit the P25's activity significantly. Moreover, in this case the polyhedral Pt nanoparticles performed slightly better than the Pt nanospheres (79% vs. 74% of degraded MO).

The Pt-containing P25 nanocomposites degraded the whole amount of oxalic acid available. As the oxalic acid molecules are

quite easily adsorbed onto the surface of the photocatalyst, they have the opportunity to react with the photogenerated holes (further enhancing the charge separation [58]). This means that the rate-determining step is on the hole side of the phenomenon, which is independent of the shape of the platinum. Also, no difference was observed in the orientation of the degradation curves, meaning that the kinetics of the oxalic acid degradation is also independent of the platinum nanocrystals' shape.

However, oxalic acid can be used as a sacrificial agent during photocatalytic hydrogen generation experiments [59], in which the rate-determining step could be the hydrogen reduction process. This can be a shape-dependent reaction, as it is very well known that electron transfer processes on different-shaped platinum nanoparticles occur differently [48].

3.5.4. Photocatalytic H_2 production efficiency

As already shown in the previous section, the shape of the platinum nanoparticles is crucial for the photocatalytic activity, influencing the different types of commercial titanias in different ways. Consequently, it was expected that a similar effect should be observable in the case of photocatalytic hydrogen production.

Indeed, as is shown in Fig. 10, the composites containing spherical Pt nanoparticles (P25-Pt(s), AA-Pt(s), AR-Pt(s)) were much more efficient in photocatalytic hydrogen production than the corresponding polyhedral platinum-containing composites (P25-

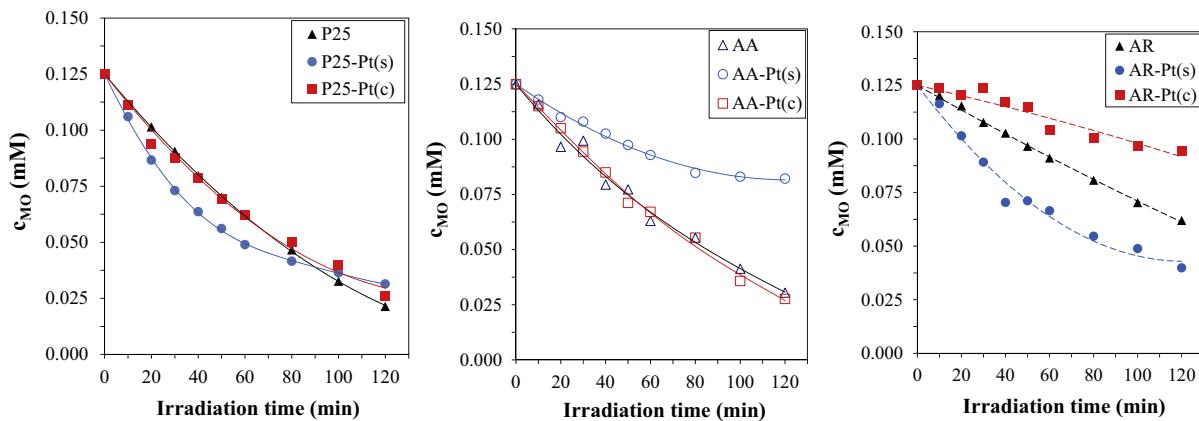


Fig. 8. The MO degradation curves for the different Pt-containing composites.

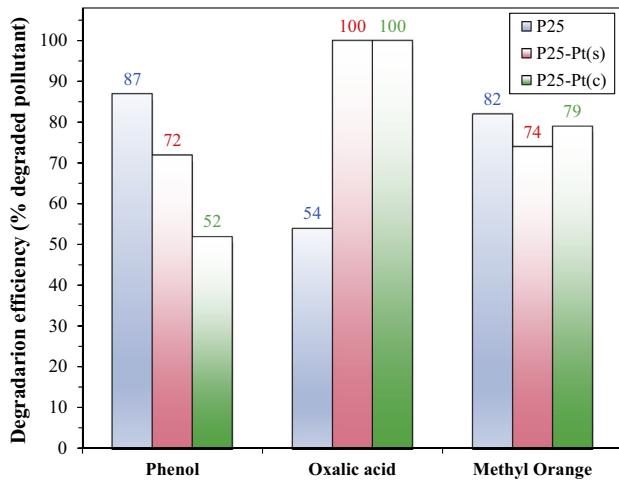
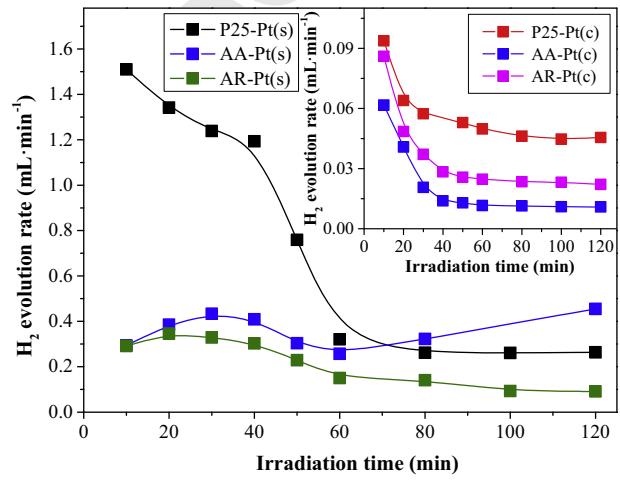


Fig. 9. Comparison of the degradation yields achieved for different substrates by P25 and P25-based composites.

Fig. 10. The hydrogen evolution rates calculated under standard conditions -25°C and atmospheric pressure for the studied nanocomposites.

Pt(c), AA-Pt(c), AR-Pt(c)), while in the case of bare commercial titanias no H_2 evolution was detected, as expected. A large activity difference was also visible in the amount of H_2 produced (Table 1) during the experimental run (2 h): 22.8–78 mL^3 of H_2 (composites with Pt(s)) vs. 2.2–6.2 mL^4 of H_2 (composites with Pt(c)).

The reasons for the obtained H_2 production efficiency values were multifold:

- **P25-based composites.** Although an activity decrease was observed in the case of phenol degradation, the fact that phenol is a poorly adsorbing substrate should be also taken into consideration, while oxalic acid adsorbs quite efficiently onto the surface of P25 [46,60]. The adsorption of oxalic acid overcompensates for the loss originated from the electron transfer process in the case of spherical Pt particles (P25-Pt(s)), but it is not sufficient to overcome the charge transfer barrier raised by the polyhedral particles (P25-Pt(c)).
- **AA-based composites.** With a relatively small surface area, oxalic acid adsorption is limited [50]; thus at first sight an insignificant H_2 production yield was expected. However, due to the

large number of available charges on the surface of AA-Pt(s), a fair amount of H_2 is produced. In the case of AA-Pt(c)⁵ the already mentioned charge barrier blocks the whole process.

- **AR-based composites.** The situation was similar to the one discussed in the photocatalytic degradation of phenol, namely that in the case of Pt(c) there are the less reactive (100) facets, while spherical Pt particles possess also very reactive high-index crystal planes. It should be noted that in the case of AR oxalic acid adsorption is nearly nonexistent due to the large crystal size (similar to AA).

3.6. Activity, structure, and morphology – the relationship between them

In the previous sections the authors listed several observations regarding the morphology dependence of photocatalytic activity and H_2 production. In some cases, preliminary explanations were provided in order to give initial insight on the phenomenology of the process. In order to clarify the details, structure-morphology-activity correlations are discussed below.

³ Calculated under standard conditions -25°C and atmospheric pressure.

⁴ Calculated under standard conditions -25°C and atmospheric pressure.

⁵ This effect is somewhat confusing, because the presence of polyhedral Pt on the surface of AA was beneficial. The issue needs further investigation.

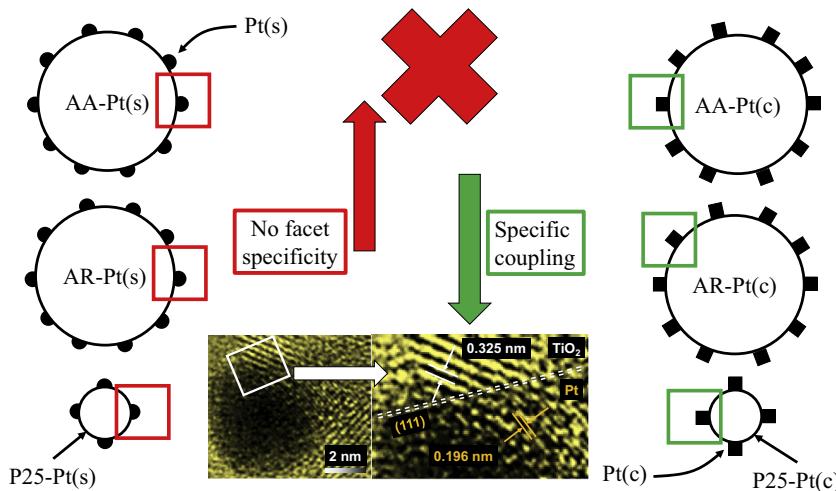


Fig. 11. Schematic representation and HRTEM evidence of the contact between Pt and TiO₂. The 0.236 nm lattice fringes correspond to the interplanar distance of rutile TiO₂ (001), while the 0.196 lattice fringes are equal to the interplanar distance of Pt (200).

596 3.6.1. The correlation between Pt(s) and TiO₂ vs. Pt(c) and TiO₂ – 597 insights on the generally higher activity of TiO₂-Pt(s) composites for 598 phenol degradation

599 One of the reasons for the generally lower activity of Pt(c)- vs.
600 Pt(s)-containing composites was the nature of the contact between
601 the commercial titania and the Pt nanoparticles. As the Pt
602 cubooctahedral contain a significant number of low-indexed facets,
603 such as (001), (010), and (100), electron transfer is inhibited com-
604 pared to that on the spherical particles that contain a large number
605 of high-indexed facets, which facilitate electron transfer [48]. It
606 was already shown that without shape control, the contact
607 between a metal and a semiconductor is realized with several crys-
608 tal facets of the metal (some of them are high-indexed ones) [61].
609 However, the contact between a shape-tailored noble metal nano-
610 particle and base catalyst should differ significantly, as shown in
611 **Fig. 11**. It can be seen that these Pt cubooctahedral particles can con-
612 nect to the TiO₂ only through their specific crystallographic planes
613 available (in the indicated example Pt (001) is one of the intercon-
614 necting facet). This observation demonstrates that when electron
615 transfer occurs from TiO₂ to Pt(c), the electron has to pass through
616 one of the mentioned facets that inhibit electron transfer. This
617 explains the generally lower activity of the Pt(c)-containing com-
618 posites vs. the Pt(s)-containing ones.

619 3.6.2. The contact number as determining factor in the photocatalytic 620 activity – AA- and AR- vs. P25-based composites

621 A major issue that was raised in this paper was the inferior
622 activity of bare AA and AR compared to their platinized versions
623 (AA-Pt(c), AA-Pt(s), AR-Pt(c), and AR-Pt(s)). An intriguing question
624 was also the reversed situation in the case of P25, where the bare
625 photocatalyst was more active than Pt-containing P25-based com-
626 posites in the case of phenol degradation.

627 Although, in each of the composite materials, the Pt content was
628 set/determined to be 1 wt.%, the TiO₂ crystal size is significantly
629 larger in AA and AR (nanocrystals, with $d > 150$ nm) compared to
630 P25 ($d = 25$ –40 nm). This means that the same amount of Pt nano-
631 particles is distributed differently among the AA, AR, and P25
632 nanocrystals.

633 For this reason, the following approaches were considered
634 (to facilitate the mathematical background of the estimation) to
635 determine the ratio of the TiO₂ and Pt nanocrystals: d_{AA} or
636 $AR = 150$ nm, $d_{P25} = 30$ nm, TiO₂ geometry – spherical, $d_{Pt} = 5$ nm,
637 $\rho_{TiO2} = 4.23$ g cm⁻³, $\rho_{Pt} = 21.45$ g cm⁻³, amount of TiO₂ 0.99 g

and 0.01 g Pt. The equation to evaluate the number of TiO₂ or
Pt is

$$N_p = \frac{3m_p}{4\pi\rho_p \left(\frac{d_p}{2}\right)^3}, \quad (1)$$

(details regarding this equation can be found in [Supporting information](#)), where N_p is the number of particles, m_p is the total mass of the particles, and ρ is the density of the chosen material, while d_p is the diameter of a single nanocrystal. The ratio of the nanocrystals can be estimated by calculating the ratio between $N_{p, Pt}$ and $N_{p, TiO2}$. The following overall particle ratio numbers were obtained:

- (i) In the case of AA- and AR-based composites: 1 TiO₂ nanopar-
ticle – 7 Pt nanoparticles.
- (ii) In the case of P25 based composites: 10 TiO₂ nanoparticles –
4 Pt nanoparticles.

The above listed two points mean that the photogenerated elec-
trons at the surfaces of AA and AR can be easily conducted away by
the 7 Pt particles available, while this advantage cannot be
assumed in the case of P25, where each third TiO₂ particle has a
single available Pt nanocrystal. Additionally, at larger crystal size
domains, the average electron conductance of TiO₂ can be several
times higher [62], suggesting that the electrons can also be more
efficiently transported through the entire TiO₂ nanoparticle. This
observation reinforces even more the efficient charge transfer
possibilities in the AA- and AR-based composites.

665 3.6.3. The nature of the chosen pollutant: in which case is the shape 666 tailoring important?

Phenol In the case of this pollutant the adsorption on the titania
668 surface is minimal [63]. Hence, the generated OH radicals are
669 responsible for the degradation process (also shown by the
670 large number of hydroxylated degradation intermediates). The
671 most probable factors responsible for the degradation of this
672 compound were already discussed in the previous two sections.
Oxalic acid This organic compound behaves differently than
673 phenol. It can adsorb to the surface of the titania extremely
674 well, which is why the chosen concentration for oxalic acid
675 was 10 times higher (at lower concentration the adsorption
676 can be a competitive process to photodegradation and the
677 two cannot easily be distinguished) [64].

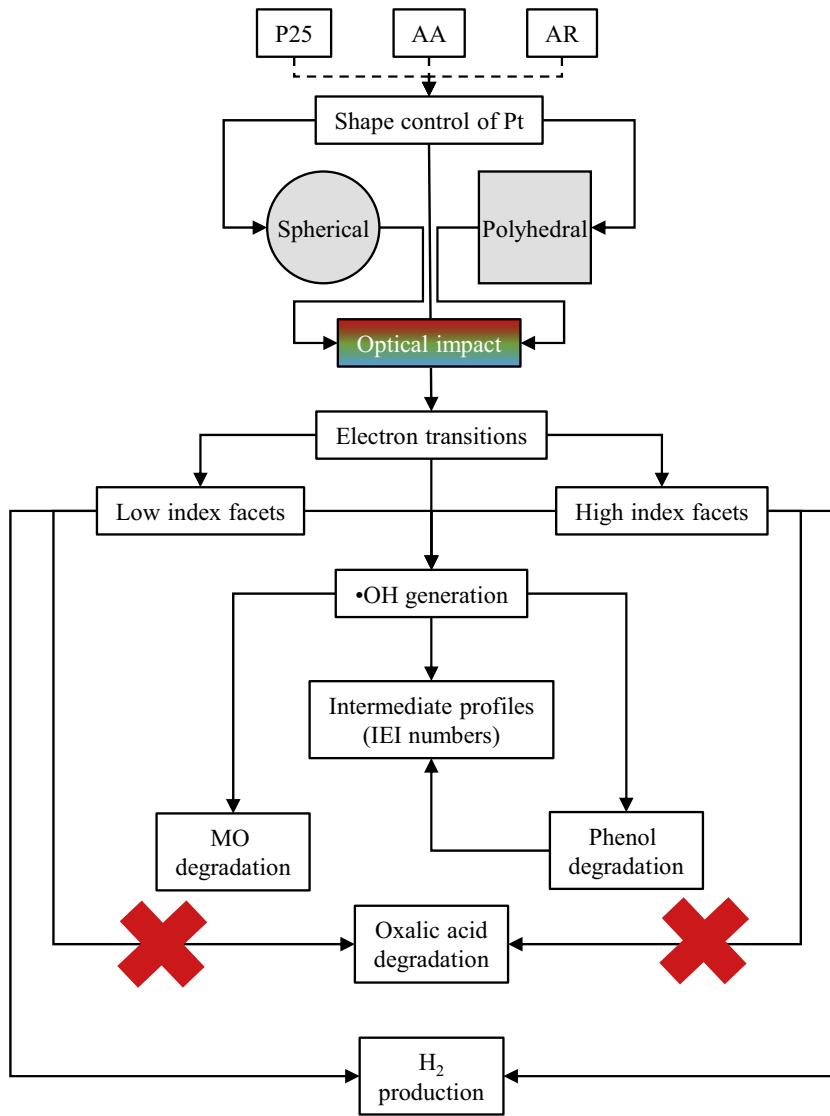


Fig. 12. A schematic overview of the impact of Pt shape tailoring.

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As oxalic acid is a hole scavenger, the electrons are efficiently separated. Since the electron flux is significantly higher than in the case of phenol (due to the holes trapped by the oxalic acid), the higher potential gradient between the Pt and TiO_2 nanoparticles overcomes the energetic barrier provided by the low-indexed crystallographic planes of Pt. Hence, the P25-Pt(s) and P25-Pt(c) are extremely efficient, and both of them show higher degradation rates than the bare P25. In this case the morphology of the Pt does not play any role.

690 3.6.4. Unclear aspects: signs regarding new experimental horizons

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Methyl orange degradation Phenol and oxalic acid are relatively simple pollutants, with predictable behavior. However, MO is a more sophisticated molecule; hence several factors can intervene in the achieved degradation efficiency, such as selective adsorption of the intermediates (this is highly possible, as several functional groups are present in the molecule: azo group, sulfonate, amine group). Thus, the observed efficiency order of the composites cannot be directly related just to the geometry of the Pt nanoparticles. However, it shows once again that shape-tailoring-driven activity tuning is more complicated than it looks at first sight.

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presence of the (100) crystal plane of Pt (in AR-Pt(c)) blocked the beneficial effect of Pt itself (of charge separation), while remaining uninfluential on the photocatalytic activity finally registered. When high-indexed crystal planes appear for Pt (AR-Pt(s)), the enhancement is again readily observable.

The latter issue was also valid for photocatalytic H₂ production experiments. All the composite materials that possessed spherical Pt nanoparticles were much more efficient than their corresponding composites with polyhedral Pt.

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Appendix A. Supplementary material

Supplementary data associated with this article can be found, in the online version, at <http://dx.doi.org/10.1016/j.jcat.2015.02.008>.

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