



Light-field-driven current control in solids with pJ-level laser pulses at 80 MHz repetition rate

VÁCLAV HANUS,^{1,*} VIKTÓRIA CSAJBÓK,¹ ZSUZSANNA PÁPA,^{1,2} JUDIT BUDAI,² ZSUZSANNA MÁRTON,² GELLÉRT ZSOLT KISS,¹ PÉTER SÁNDOR,¹ PALLABI PAUL,³ ADRIANA SZEGHALMI,^{3,4} ZILONG WANG,⁵ BORIS BERGUES,^{5,6} MATTHIAS F. KLING,^{5,6} GYÖRGY MOLNÁR,⁷ JÁNOS VOLK,⁷ AND PÉTER DOMBI^{1,2}

¹Wigner Research Centre for Physics, Konkoly-Thege M. u. 29-33, Budapest 1121, Hungary

²ELI-ALPS, ELI-HU Non-Profit Ltd., Wolfgang Sandner u. 3, Szeged 6728, Hungary

³Institute of Applied Physics, Abbe Center of Photonics, Friedrich Schiller University Jena, Albert-Einstein-Str. 15, 07745 Jena, Germany

⁴Fraunhofer Institute for Applied Optics and Precision Engineering, Centre of Excellence in Photonics, Albert-Einstein-Str. 7, 07745 Jena, Germany

⁵Physics Department, Ludwig-Maximilians-Universität Munich, Am Coulombwall 1, 85748 Garching, Germany

⁶Max Planck Institute of Quantum Optics, Hans-Kopfermann-Str. 1, 85748 Garching, Germany

⁷Centre for Energy Research, Institute of Technical Physics and Materials Science, Konkoly-Thege M. u. 29-33, Budapest 1121, Hungary

*Corresponding author: hanus.vaclav@wigner.hu

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Future PHz electronic devices may be able to perform operations on few-femtosecond time-scales. Such devices are based on the ability to control currents induced by intense few-cycle laser pulses. Investigations of this control scheme have been based on complex, amplified laser systems, typically delivering mJ or sub-mJ-level laser pulses, limiting the achievable clock rate to the kHz regime. Here, we demonstrate transient metallization and lightwave-driven current control with 300-pJ laser pulses at 80 MHz repetition rate in dielectric media (HfO₂ and fused silica), and the wide-bandgap semiconductor GaN. We determine the field strength dependence of optically induced currents in these media. Supported by a theoretical model, we show scaling behaviors that will be instrumental in the construction of PHz electronic devices. © 2021 Optical Society of America under the terms of the [OSA Open Access Publishing Agreement](#)

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1. INTRODUCTION

It is a substantial challenge to construct solid-state photonic devices capable of performing operations on the timescale of the optical cycles of visible or infrared lightwaves. Such devices are highly desired for petahertz (PHz) electronics applications [1,2] or to construct new diagnostic instruments for few-cycle laser pulses. A PHz electronics device can be based on the effect of transient metallization of wide-bandgap materials or, alternatively, on the ultrafast electron transport across vacuum nanojunctions in field-enhancing nanoarchitectures [3–5]. The effect of metallization has been already demonstrated in dielectrics [6–8] such as SiO₂, CaF₂, or Al₂O₃, and semiconductors [9,10] such as GaN or graphene [11]. Unlike nanojunctions, it has the potential to circumvent technological challenges such as reliable nanojunction fabrication or the requirement of off-resonant plasmonic nanostructures [12] to support PHz bandwidth. Few-cycle laser-pulse-induced femtosecond transient metallization has proven to be a promising phenomenon, thanks to its unprecedentedly fast electronic response [13]. Unlike the competitive phenomenon of current generation via interference of two quantum pathways in the transitions between the valence and conduction band [14–16], the transient metallization is considered *not* to produce real carriers in the conduction band [2,13], which is advantageous in terms of a

large supporting bandwidth and low dissipation of the deposited energy. Optically induced currents in materials, however, are limited to a narrow parameter range of the laser pulses, since strict requirements must be met with respect to intensity and pulse length. Even for few-cycle laser pulses where the damage threshold is increased with respect to longer pulses, the peak intensity is clamped by optical damage phenomena. This limits the applicable intensity range to a narrow domain. In spite of these limitations, first applications of the transient metallization effect have already been demonstrated in the development of carrier-envelope phase (CEP) detectors [17] or PHz waveform samplers [18]. These demonstrations also involved control of optically induced currents with the carrier-envelope phase of the laser pulses, as a hallmark for the strong-field nature of this process and its capability to support PHz bandwidths [1].

The first observation [6] indicated that the phenomenon of the transient metallization in dielectrics is limited to fields of about 1 V/Å and higher. These fields have been typically provided by mJ-class laser systems with very low, ~ kHz repetition rates. The requirement of high field strength to trigger the phenomenon in dielectrics would hinder the miniaturization and mass applicability of potential devices. High-pulse-energy laser systems are bulky and

have a low repetition rate, enabling only proof-of-principle experiments. An important step toward device-oriented applications was very recently made by Langer *et al.*, where current control in GaN was demonstrated with a parametric amplifier system delivering 75 nJ pulses with a 200 kHz repetition rate [6]. Even in this case, however, the complexity of the light source seems prohibitive for future applications. The demonstration of optical current control purely in the bulk of a dielectric material with pJ-class, high-repetition-rate, oscillator-based laser pulses is thus highly desirable but still lacking.

Here, we fill this gap and report on the transient metallization and CEP-driven current control induced in a compact setup at a 80 MHz repetition rate with pJ-level pulses in GaN, HfO₂, and SiO₂. We demonstrate current control with a two orders of magnitude higher repetition rate and also a two orders of magnitude lower pulse energy compared to previous work. In addition, we prove that pure insulators can be used with nJ- and pJ-level laser pulses. We determine the field strength dependence of the optically generated currents and compare the current control phenomenon in the range of bandgaps between 3.5 and 9 eV. In addition, our results on pure insulators permit the movement of current control technology toward GHz repetition-rate femtosecond sources, a clock rate that is already competitive for ultrafast electronics applications.

2. EXPERIMENTAL SETUP FOR CURRENT CONTROL

We investigated the electric signal from different semiconductor and dielectric materials illuminated by laser pulses under ambient laboratory conditions in a simple geometry where a laser beam is focused between two gold electrodes into an active layer of material under investigation. For more details on the structure of

our samples, Fig. 1(a) shows the geometry of the target and the illumination by few-cycle pulses. The laser beam is provided by a Ti:Sapphire laser oscillator (Venteon Laser Technologies) that provides an octave spanning spectrum [19], as shown in Fig. 1(b). The carrier-envelope offset frequency (f_{CEO}) of the pulse train can be stabilized to an arbitrary value. The oscillator emits laser pulses of 2.5 nJ at a repetition rate of 80 MHz. The duration of the pulses was characterized with a d-scan [20] yielding 5.4 fs full width at half-maximum (FWHM) of the intensity envelope. In Fig. 1(c), we show the d-scan performed in situ. To ensure the shortest on-target duration, the pulses were pre-chirped with dispersive mirrors. The pulse chirp was then fine-tuned using a pair of broadband antireflection-coated fused silica glass wedges. The maximum yield of the second harmonic signal obtained during the d-scan indicated an optimal compression of the pulses at the place of the sample. The glass wedges also allow continuously change of the CEP of the pulses impinging on the target. Details on the d-scan trace formation and reconstruction are provided in Supplement 1.

Coupling of the laser beam to the current control device is realized by tight focusing with off-axis parabolas. To achieve high on-target intensity, the beam was expanded with a telescope to a diameter of 7 mm prior to focusing. By using focusing parabolas with different focal lengths, we achieved two on-target focal spot sizes: 1.8 and 2.3 μm , measured as FWHM of the intensity profile. The focal spot sizes were characterized with a knife-edge scan. This scheme supported intensities up to $6 \times 10^{12} \text{ W cm}^{-2}$ with corresponding field strengths of up to 0.5 V/Å.

We demonstrated the high-repetition-rate current control effect in three materials: (i) epitaxially grown GaN (with a bandgap of 3.5 eV and a layer thickness of 2 μm) deposited onto a sapphire plate; (ii) HfO₂ (with a bandgap of 5.9 eV and layer thickness of 270 nm, chosen as a compromise between fabrication difficulty

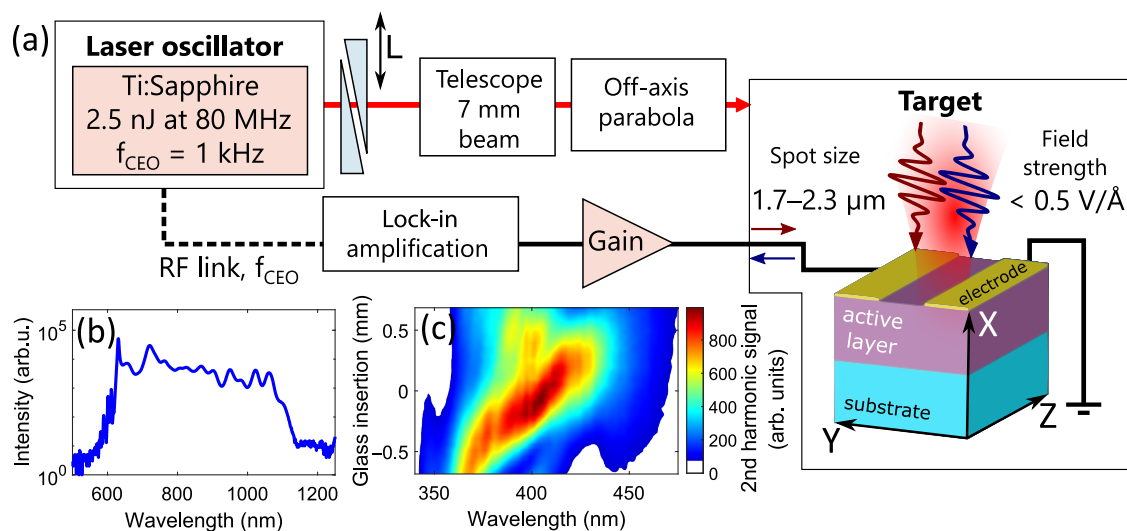


Fig. 1. (a) Scheme of the experimental setup. The laser beam from the oscillator is guided through a pair of fused silica dispersion compensation wedges. Then the beam is expanded by a telescope to 7 mm diameter and focused onto the target by an off-axis parabola. The target is composed of a transparent substrate and an active layer of the investigated material (in case of SiO₂, only the substrate is present). On top of the active layer, electrodes are patterned, forming a gap into which the laser beam is focused. Two optical waveforms (red and blue) are shown for two different (opposite) CEP values. The electrodes are connected with a pre-amplifier and lock-in amplifier. The current direction, shown with dark red arrows and dark blue arrows, oscillates as the CEP of pulses in the pulse train changes with the f_{CEO} frequency. The lock-in amplifier is referenced via a phase-locked radiofrequency link to f_{CEO} of the laser pulse train. (b) Spectrum of the laser output acquired during the phase-locked operation. (c) D-scan trace acquired during the phase-locked operation showing the second harmonic signal created in a BBO crystal (replacing the current control target) as a function of wavelength and amount of glass insertion. Zero glass insertion is calibrated to the position of maximum of the signal's marginal integrated over the wavelength axis.

and maximization of volume of the investigated material in the sample) deposited by atomic layer deposition onto a fused silica substrate (see Supplement 1 for fabrication details); and (iii) in a bulk fused silica plate (SiO_2 with a bandgap of 9.0 eV). Gold electrodes were patterned onto the substrates with electron beam lithography. The electrodes were shaped into a step-like structure so that junctions with different gap widths (0.6, 1.2, 1.7, 2.3, and 2.9 μm) are accessible within the range of high-resolution linear piezo stage, as shown by the contours of the electrodes' edges in Fig. 2. Each gap has a length of 5 μm providing that the laser focal spot can be well contained within one junction of a certain size.

The ultrafast current induced during the passage of the femtosecond pulse through the material leads to an accumulation of a charge on the attached electrodes. This accumulated charge then drives a slow current in the connected circuit that is then amplified by a current preamplifier (DLPCA-200, Femto Messtechnik GmbH, Berlin, Germany). Subsequently, the signal is analyzed with a low-frequency, lock-in amplifier (SRS SR830) referenced to the carrier-envelope offset frequency, f_{ceo} . As the CEP of the pulse is related to f_{ceo} according to $\phi_{\text{CE}} = 2\pi f_{\text{ceo}}t$, it is the carrier-envelope offset frequency at which we expect the change of polarity of accumulated charge on the electrodes. We chose to lock f_{ceo} to 1 kHz as the highest value of the frequency is limited by the bandwidth of the preamplifier, while the chosen value should be high enough to provide a stable signal. The lock-in amplifier measures the voltage, V , the output of the preamplifier can be converted to current, J , by dividing by the gain factor, G , of the preamplifier: $J = G^{-1}V$. Finally, two quantities representing the current are retrieved: in-phase component J_{cos} and quadrature J_{sin} , which represent the rms magnitudes of the Fourier components at the reference frequency f_{ceo} . The total current amplitude J_0 and its phase ϕ_J are then calculated from these quantities as $J_0^2 = (J_{\text{cos}}^2 + J_{\text{sin}}^2)$ and $\phi_J = \text{atan}(J_{\text{sin}}/J_{\text{cos}})$. The preamplification factor G was 10^8 V/A for GaN and 10^9 V/A for HfO_2 and SiO_2 , since GaN delivered higher signals than the other targets. The spatial profile of J_0 from the GaN target was retrieved by 3D scanning the position of the sample in the focal plane and laser propagation direction. In Fig. 2(a), we show this profile for the point of the highest signal in the laser propagation direction obtained for laser pulses of 1 nJ

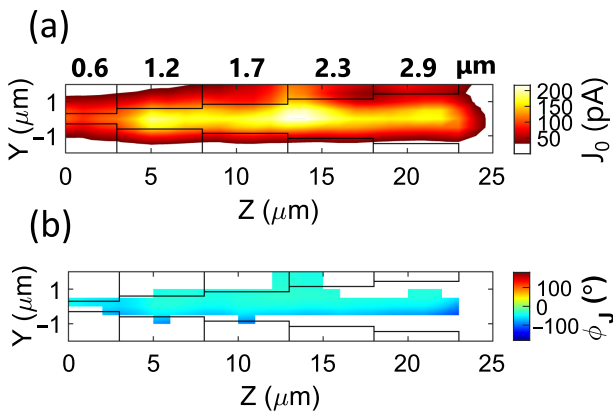


Fig. 2. (a) CEP sensitive current signal, J_0 , obtained with 1 nJ pulses from the GaN target as a function of its position in the focal plane of the laser beam for a focal spot size of 1.8 μm and a peak intensity of 3.2×10^{12} W/cm 2 . Black contours represent the edges of specially designed, step-like electrodes, on the surface of the target. The widths of the gaps between the electrodes are labeled in bold type. (b) Measured phase ϕ_J of the current signal is displayed for the area where the current value J_0 was larger than half of the maximum current in (a).

with corresponding peak intensity of 3.2×10^{12} W/cm 2 . One can observe some weak dependence of the signal strength and the junction width. Interestingly, the highest signal is obtained for widths of 1.7 and 2.3 μm , close to the applied focal spot size of 1.8 μm . Despite the varying signal strength with the position on the target, the detected phase ϕ_J does not change significantly. To illustrate this, we show in Fig. 2(b) the phase ϕ_J for the position where high signal J_0 was measured. The region of high signal was selected as all positions having a higher signal than half of the maximum detected signal in Fig. 2(a). The standard deviation of the measured phase is only 18° with a mean value of -49° .

3. OPTICAL CURRENT CONTROL WITH pJ-LEVEL PULSES

We observed CEP-dependent currents with the high-repetition rate driver in our semiconductor and dielectric media: GaN, HfO_2 , and SiO_2 . To crosscheck that the signal is CEP sensitive, we performed a scan with wedge glass insertion into the laser beam [21], as shown by glass insertion parameter L in Fig. 1(a). As the wedge is moved and L changes, the path of the laser beam in the wedge material, d , also changes. Consequently, as the CEP of pulses including the wedge effect is defined as $\phi_{\text{CE}} = 2\pi f_{\text{ceo}}t + \phi(d)$, the quantities J_{cos} and J_{sin} oscillate with the phase offset $\phi(d)$ introduced by the dispersion in the material of the wedges, as shown in Figs. 3(a)–3(c). This clearly proves the few-cycle nature and CEP sensitivity of the optically generated current.

The change of the path length in the wedge material also affects the pulse duration. Therefore, the current strength J_0 can be expected to be the highest for the shortest on-target pulse duration. Indeed, we observe peaking of the current strength J_0 for a given wedge insertion, as shown in Figs. 3(e)–3(g). We relate the trace of

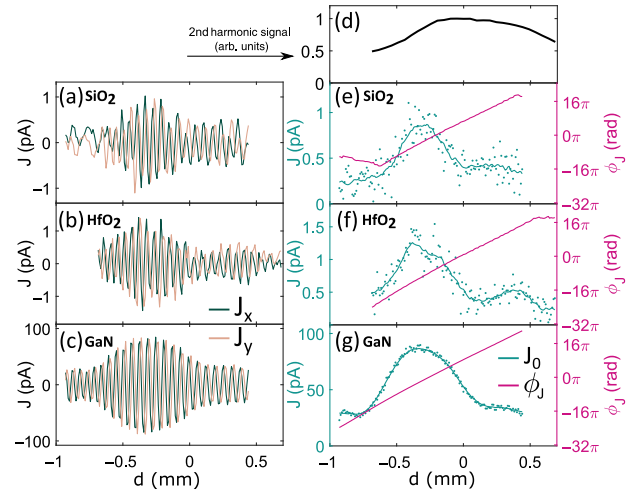


Fig. 3. (a)–(c) J_x (green) and J_y (beige) are the in-phase and quadrature components of the current, respectively, at the frequency f_{ceo} . The signal is shown as a function of the amount of glass of a wedge inserted to the beam, d , for targets made from SiO_2 , HfO_2 , and GaN from top to bottom. Wedge scans were done with following focal spot sizes and laser peak intensities: 2.3 μm , 3.7×10^{12} W/cm 2 (SiO_2); 1.8 μm , 4.9×10^{12} W/cm 2 (HfO_2); and 2.3 μm , 1.4×10^{12} W/cm 2 (GaN). The second column (e)–(g) shows the measured magnitude of current J_0 (green) and its measured phase ϕ_J (magenta) for the same targets. The green line is a running average of the green measured points. The zero point of the glass insertion was calibrated to the maximum of the marginal (d) of the d -scan from Fig. 1(c).

the current strength to the d-scan trace performed at the position of the target. In Fig. 3(d), we show the marginal of the d-scan trace from Fig. 1(c) (i.e., the integrated trace over the wavelength dimension). One can see in the current trace that, as the dispersion changes, the pulse is becoming compressed and simultaneously the intensity increases. The second harmonic signal of the d-scan also increases until it reaches a maximum; from then on, the pulse is stretched and its maximum intensity decreases again. It is noteworthy that this marginal is a result of a process depending on the second power of the electric field strength and forms a shallow peak in the curve shape. The current signals in Figs. 3(e)–3(g), in contrast, rise in a steeper manner. This clearly proves that the current signal is a result of a higher-order process compared to the second harmonic generation in our pulse diagnostic instrument. The field strength dependence of the current signal is further investigated and analyzed below. Additionally, the shape of J_0 as a function of glass insertion d is not completely symmetrical as the signals form a plateau right from the main peak (at $d \sim 0.25$ mm). We attribute this to the different pulse shapes obtained for positive chirping versus those for negative chirping by the glass wedges. Additionally, since the pulse is generally not Gaussian, some local pulse duration minima appear as a function of d and these minima enhance the peak intensity of the laser pulse for certain d . (See Fig. S2 in Supplement 1 to see the pulse shape as a function of d .)

An important characteristic is the pulse intensity or equivalently the field amplitude (E_0) dependence of the current signal because the order of nonlinearity of the dependence can provide insight about the excitation process taking place between the valence and conduction bands of the given material. Such an approach is coherent with previous studies that related the current control to the process of coherent superposition of wave packets originating from different excitation pathways [10,16]. Similar concepts are also widely applied in the multiphoton ionization of gases [22] as well as in solid-state media [16,23]. Therefore, we performed a laser power scan of the current signal from GaN, HfO₂, and SiO₂ media, as shown in Fig. 4. The average power of the pulse train was measured right before the focusing parabola with a thermopile power meter head. The pulse energy ε and the amplitude of the field amplitude E_0 in the target medium was calculated from the measured focal spot size and the d-scan-reconstructed pulse duration. The finite reflectivity of the parabolic mirror and Fresnel reflection from the surface of the sample were taken into account. Positioning of the target into the focus was performed by manual search for the maximum current signal while adjusting the xyz coordinates of the target with submicron-precision actuators.

Multiple power scans were performed for each material, out of which the average value of J_0 is displayed in Fig. 4 together with error bars representing the standard deviation across the set of measurements. Along with the magnitude of current, we also measured the phase of the signal. The phase was stable during the time of the measurement, which took only a few minutes. As it can be seen, the increase of signal with the field strength is different for the cases of GaN, HfO₂, and SiO₂. This could be related to the order of nonlinearity of the current generation process involved, which depends on the bandgap E_g and the photon energy. Assuming excitation of *real* carriers and the current originating from the interference of multiple multiphoton pathways, the expected relationship [10] between the signal and the E-field would hold: $J_0 \propto E_0^{2N+1}$, where $N = \frac{E_g}{\hbar\omega}$ is the number of photons needed to populate the conduction band. Considering the range of photon

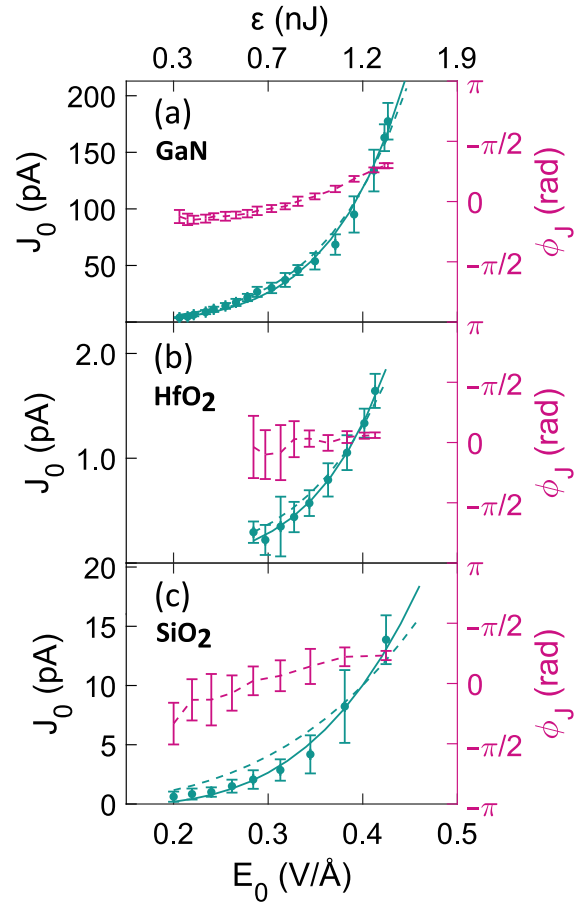


Fig. 4. Dependence of the total current amplitude, J_0 (green), and its phase, ϕ_J (magenta), on the electric field strength, E_0 , measured at the reference frequency f_{cen} . It is shown for targets made from (a) GaN, (b) HfO₂, and (c) SiO₂. All targets were measured with a focal spot size of 1.8 μm . Abscissa on the top serves to show nonlinear scaling with pulse energy. A vertical offset was applied to the phase to center the mean of the phase to zero. Dashed green curves were obtained by fitting according to the model described in Eq. (1). The solid line was obtained using the same model extended with a correction, taking into account the expansion of the surface that contributes to the current generation as the intensity is increased.

energies included in the ultrabroadband laser spectrum, as shown in Fig. 1(b), one can consider multiphoton transitions between the valence and conduction band with a multiplicity of two to three photons in case of GaN and five to nine photons in the SiO₂ case. Hence, the exponent of the curve would be between five and seven for GaN and at least 11 for SiO₂. Power fits $J_0 \propto E_0^q$ applied to the results (not shown) provide $q = 6.3$ and 5.6 for these two materials, respectively. Although this would explain the slope in the GaN measurement, it cannot explain the data for SiO₂, where the exponent differs by more than five. This means that perturbation theory considering multiphoton transitions and exciting real carriers cannot be effectively applied to the range of parameters and materials under investigation, especially not for SiO₂; thus, a different model must be applied.

For a better description of the observed field-strength dependencies for all investigated materials, we turned toward Khurgin's model, as described in [24]. This model was applied successfully for current control results in GaN [9]. In this model, the current is a consequence of a photogalvanic effect in a $\chi^{(3)}$ -type medium

via nonlinear conductivity. The three photon interference $[\omega_1, \omega_2, -(\omega_1 + \omega_2)]$ generates the DC current; as soon as the participating photons have energy lower than the half of the band gap, no real charge carriers in the conduction band are populated. Conversely, the concept of virtual carriers is introduced because the interband transitions also are featured in the mathematical description. We compared our experimentally measured photo-induced $J_0(E_0)$ currents with the theoretical curves obtained by applying Khurgin's model. Within this model, we assume a femtosecond linearly polarized Gaussian pulse with a carrier frequency below bandgap frequency. The electric field of the pulse is defined as $E(t) = E_0 \exp(-2 \ln 2 t^2 / \tau_p^2) \cos(\omega_0 t + \phi_{ce})$. The laser pulse induces an asymmetric distribution of virtual carriers in the conduction and valence bands in the dielectric or semiconductor. After the passage of the ultrashort laser pulse, a net charge Q is accumulated in the medium and

$$Q(E_0) = \varepsilon_0 A_{\text{eff}} \chi^{(3)} \sum_{m=1}^{m_{\text{max}}} \left(\frac{\chi^{(3)}}{\chi^{(1)}} \right)^{m-1} \langle a^{2m+1} \rangle E_0^{2m+1}, \quad (1)$$

which flows to the circuit through the electrodes and produces a current $J_0 \sim Q(E_0) / \tau_{\text{RC}}$, with τ_{RC} being an unknown characteristic time of the circuit. A_{eff} is a parameter that takes into account the efficient surface area of the target. According to this formula, the dimensionless $\langle a^{2m+1} \rangle = \omega_0 \int \langle a(t)^{2m+1} \rangle dt$ odd-power moments of the normalized vector potential [i.e., $a(t) = (\omega_0 / E_0) \int_{-\infty}^t E(t') dt'$] were numerically evaluated for the laser pulse parameters $\tau_p = 5.5$ fs and $\omega_0 = 2\pi c / \lambda$, with $\lambda = 800$ nm. A numerical convergence test was performed to obtain m_{max} and, for the considered field strengths, it yielded convergent results already for $m_{\text{max}} = 5$ for each target material. For the first-order susceptibility, we considered $\chi^{(1)} = n^2 - 1$, with n being the refractive index of the target material. Finally, a fitting was performed to find values for $A_{\text{eff}} / \tau_{\text{RC}}$ using table values for third-order susceptibility $\chi^{(3)}$ for GaN: $\chi^{(3)} = 2.1 \times 10^{-20} \text{ m}^2 / \text{V}^2$ [25] and SiO₂: $\chi^{(3)} = 2 \times 10^{-22} \text{ m}^2 / \text{V}^2$ [26]. An approximate value of the third-order susceptibility for HfO₂ was obtained from Miller's relation [27] as $\chi^{(3)} = 3.5 \times 10^{-21} \text{ m}^2 / \text{V}^2$.

Results of the fits using Khurgin's model are presented with dashed lines in Fig. 4. We obtained satisfactory results for all targets. Yet, for SiO₂ the model underestimates the steepness of the curve. To rectify this discrepancy, we amended the model to take into account the effects of the size of the illuminated area on the sample. The inherent property of Khurgin's model is that it assumes constant E-field strength across the area in the focal plane. As we operate our experiment in a tight focusing regime (as opposed to previous similar studies [9,10]), the effect of the nonuniform illumination might become significant. Therefore, we added a correction factor to Eq. (1), taking into account that the area above some critical value of field strength E_{crit} increases with applied E-field amplitude E_0 . The value E_{crit} represents a threshold from where the effect starts to appear. This value also depends on the material. Having E_{crit} and the laser waist w_0 fixed, the E-field profile of the area illuminated by a Gaussian laser is given as $E(r) = E_0 \exp(-2r^2 / w_0^2)$. Hence, the radius r_c of the area contributing to the current generation, defined as $E > E_{\text{crit}}$, is obtained from $E_{\text{crit}} = E(r_c)$. From there, it can be deduced that the effective area scales with the field amplitude according to $A_{\text{eff}}(E_0) \propto \pi w_0^2 \ln E_0 / E_{\text{crit}}$. Considering this, we amended the fits obtained from the model in Eq. (1), as shown by the solid

lines in Fig. 4. E_{crit} values were taken as new parameters for the fit and their values were determined to be 0.06, 0.12, and 0.17 V/Å for GaN, HfO₂, and SiO₂, respectively. Notice that this sequence of critical fields corresponds to the increasing bandgap of the targets (3.5 eV, 5.9 eV, and 9.0 eV, respectively). We note that, in case of GaN, the phenomenon is observable even for lower field amplitudes than presented in Fig. 4(a). Our analysis of E_{crit} indicates that the effect can be initiated already with 0.06 V/Å, which corresponds to ~ 50 pJ of pulse energy.

With measurement curves reproduced well with the model in Fig. 4, it remains to elucidate the background of the different field dependencies observed. A different slope is particularly noticeable for the GaN and the SiO₂ samples depicted in Figs. 4(a) and 4(c), respectively. Going deeper in the details of the model from Eq. (1), we can show that the slope is determined by the material constant $\chi^{(3)}$ and its relation to the applied E_0 amplitude and the value of coefficients $\langle a^{2m+1} \rangle$. This is because $\chi^{(3)}$ can be approximately related to the atomic field E_a as $\chi^{(3)} \sim E_a^{-2}$ [24]. When applied to Eq. (1), one can see that the significance of individual terms in the expansion is given by the ratio E_0 / E_a . The closer the ratio is to one, the higher the contribution of higher-order terms. Moreover, the coefficients $\langle a^{2m+1} \rangle$ can have orders of magnitude different values. For example, for the considered pulse of 5.5 fs duration, a^7 is 150 times higher than a^3 . Analysis of the contributions of the individual terms shows that the GaN curve in Fig. 4(a) is determined mostly by terms of fifth- and seventh-order. SiO₂ is determined by the 3rd order with little contribution from the fifth order and finally HfO₂ is defined by more or less equal contributions of the third- and fifth-order terms. For more details, see Fig. S3 in Supplement 1, where the individual terms from the expansion in Eq. (1) are plotted.

Another interesting feature is the phase dependence of the signal. The measurement shows that the phase of the current has very low field amplitude dependence for all materials, as shown by the purple line in Fig. 4. From 0.2 V/Å to the maximum probed intensity about 0.4 V/Å the phase is slightly monotonically increasing with field strength. This could be a promising property for a CEP detector for online phase tagging applications [17,28] used with laser oscillators. For example, a robust phase characterization could be performed with a pulse providing an electric field strength on the GaN target between 0.2 and 0.3 V/Å where the phase change in this interval is almost flat and changes only by 12°.

Note that there are contrasting models for transient metalization presented in the literature [6,13,23,24] and obtaining a phase-sensitive current for such low field strengths in case of the SiO₂ target is a striking finding. In the originally introduced picture [6,13], the strong, but adiabatic, optical fields shift the original valence and conduction bands to create a new state of matter formed by so-called Wannier–Stark ladders. This model requires high electric fields of ~ 1 V/Å to allow transitions between the Wannier–Stark shifted valence and conduction band of directly neighboring lattice sites. The observation of the phenomenon with a lower field suggests that transitions between remote sites of the lattice are not negligible. A more recent model by Chen *et al.* [23] builds upon the interaction between the valence and conduction bands of the dielectric at the terminal crystal cells with the Fermi sea of electrons in the electrodes. Transient states are created where the electrons from the metal have enough energy (in relation to the conduction band) and they can hop to the conduction band, resulting in a current. A condition arises from this reasoning that

the process can be triggered only for fields higher than 1.4 V/Å. Unlike other models, this one even predicts opposite CEP for which the maximum current is achieved. In spite of that, our observation shows that it is possible to achieve current control well below this limit and the role of electrodes in the process still must be discussed. Apparently, understanding of the transient metallization in wide-bandgap materials is yet to be settled and new observations and measurements for a different set of parameters and materials are needed.

4. SUMMARY

In summary, we demonstrated light-field-driven control of optically induced currents in dielectrics and wide-bandgap semiconductors with unprecedentedly low, pJ-level laser pulse energy and 80 MHz repetition rate. Despite the fact that the currents were detected with slow circuitry, the directional sensitivity of the measured current to the subcycle evolution of the CEP underpins the sub-PHz bandwidth nature of the interaction between the laser and the dielectric (semiconductor) medium and electrode. Applying a theoretical model to our data, we could explain the dependence of the signal on the electric field amplitude of the laser pulse, which supports the understanding of these currents as a nonresonant phenomenon. In other words, even when real carriers cannot be produced by multiphoton transitions, virtual carriers are being generated and their distribution in momentum space, being asymmetric due to the nonzero time-integral of the laser vector potential, produces real currents detected by external circuitry. Consequently, our data indicate that current control can be obtained for pulses as weak as 50 pJ. The fact that the phase of the detected current is insensitive to the pulse energy hints at the possibility to design robust CEP detectors of laser oscillator pulses. With the results demonstrated, we believe our experiments pave the way toward ultrafast, PHz-bandwidth optoelectronic devices operational with pJ-level pulse energy being instrumental for future, real-life applications of ultrafast, integrated optoelectronics.

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Data Availability. Data underlying the results presented in this paper are not publicly available at this time but may be obtained from the authors upon reasonable request.

Supplemental document. See Supplement 1 for supporting content.

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