

Figure 1. (A) X-ray diffraction patterns of Au–Sn bimetallic NPs and the parent metals. The diffractions corresponding to the respective crystal phases in the samples are indicated by these marks: (◆) Au, (⊙) AuSn, (●) AuSn₂, (Δ) Sn. (B) TEM image and (C) HRTEM image of the Au₁Sn₂ sample. Lattice fringes are highlighted together with the corresponding crystal facet.

became more selective toward the production of hydrogen and methane.²⁶ Atomically ordered AuCu NPs selectively reduced CO₂ to CO, achieving FE of 80%; in contrast, disordered AuCu NPs favored the hydrogen evolution reaction (HER).²⁰ Au₃Cu nanocubes with vacancy defects showed an overpotential lower than that of the Au₃Cu alloy and Au NPs because dealloying tuned the *COOH/*CO binding strength and the selectivity toward CO production.²⁷ A Cu–Sn alloy resulted in the selective reduction of CO₂ to CO with a FE ≥ 90%.²⁸ Cu/SnO₂ core–shell NPs showed thickness-dependent CO₂ reduction properties: the thicker shell (1.8 nm) acted like SnO₂ NPs, and formate was the major product, whereas the thinner shell (0.8 nm) was selective toward CO formation with FE = 93%.¹² A volcano-type curve was observed between FE_{HCOOH} and bulk tin content on Ag–Sn electrocatalysts, having a bimetallic core and a SnO_x shell.⁴ Finally, the contribution of geometric and strain effects was isolated on the example of Ru–Pt NPs (Ru@Pt core–shell and RuPt alloy). The Ru@Pt NPs with high compressive strain had HER activity in alkaline conditions that was better than that of a RuPt alloy with strain-free surface.²⁹

To the best of our knowledge, the Sn–Au combination has not been employed in CO₂ reduction yet. This is indeed surprising, considering that gold and tin are both known to be good catalysts for electroreducing CO₂ to form CO and HCOOH, respectively. Here, we present the synthesis of Au–Sn bimetallic NPs with well-defined morphology and compositions and explore their composition-dependent catalytic activity toward CO₂ reduction for the first time. The formation of two value-added products (syngas and formate) was demonstrated with considerable efficiency, and their ratio was tuned by changing the composition of the intermetallic phase(s).

Bimetallic Au–Sn NPs with different nominal compositions (i.e., Au₂Sn₁, Au₁Sn₁, Au₁Sn₂, and Au₁Sn₄) were synthesized

using a two-step synthesis approach.³⁰ Detailed description of the synthesis is given in the Supporting Information, but briefly, this method relies on the chemical reduction of different amounts of tin precursor in the presence of pre-made gold NPs,³¹ acting as nucleation seeds. The crystal structure of the NPs was determined using powder X-ray diffraction (XRD) (Figure 1A). The pure gold showed peaks at $2\theta = 38.14, 44.44, 64.71, \text{ and } 77.73^\circ$ corresponding to face-centered cubic phase of gold. The bimetallic phases showed distinctly different XRD patterns, confirming that new phases were formed (i.e., no simple alloying occurred). The Au₂Sn₁ sample had additional peaks at 2θ of 23.67, 28.74, and 40.53° assigned to the hexagonal AuSn intermetallic phase. The AuSn phase became more prevalent with further increase of Sn⁴⁺ concentration (sample Au₁Sn₁), and even the formation of phase-pure AuSn was obtained (sample Au₁Sn₂). A different diffraction pattern was observed at the highest tin concentration (sample Au₁Sn₄), associated with the orthorhombic intermetallic AuSn₂ phase. The pure tin exhibited a tetragonal phase. Rietveld refinement of the diffraction patterns was carried out to quantify these trends (Figure S1 and Table S2). The morphology of the Au–Sn NPs and their monometallic counterparts was characterized by transmission electron microscopy (TEM). Gold NPs were mainly spherical and crystalline, with an average diameter of 22.4 ± 2.2 nm (Figure S2). After the incorporation of tin, the size of the NPs was $23.0 \pm 2.9, 31.8 \pm 3.9, 32.4 \pm 3.7, \text{ and } 33.0 \pm 2.5$ nm in the series of samples with growing tin content (Figures S3–S6).

The chemical nature of the NP surface was characterized by XPS. Figure S7 shows the XPS fitting of Au 4f and Sn 3d spectra for the four bimetallic samples. The amount of the partially oxidized tin species on the surface increased gradually with the total tin concentration: the sample with the highest tin content (Au₁Sn₄) exhibited only a single Sn 3d doublet at 494.81 and 486.15 eV that corresponded to Sn^{4+/2+}. A minor,

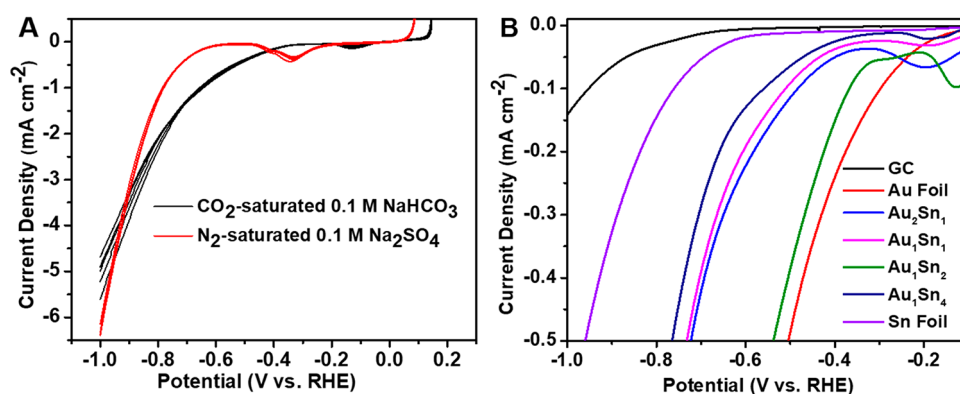


Figure 2. (A) LSV profiles of Au₁Sn₂ catalyst recorded in CO₂-saturated 0.1 mol dm⁻³ NaHCO₃ and N₂-saturated 0.1 mol dm⁻³ Na₂SO₄. Scan rate = 5 mV s⁻¹. (B) LSV profiles of Au–Sn NPs and gold and tin foils, recorded in CO₂-saturated 0.1 mol dm⁻³ NaHCO₃ stabilized after multiple cycles. The black line indicates the activity of the bare glassy carbon substrate. Scan rate = 5 mV s⁻¹. The loading was 0.35 mg cm⁻² in all cases.

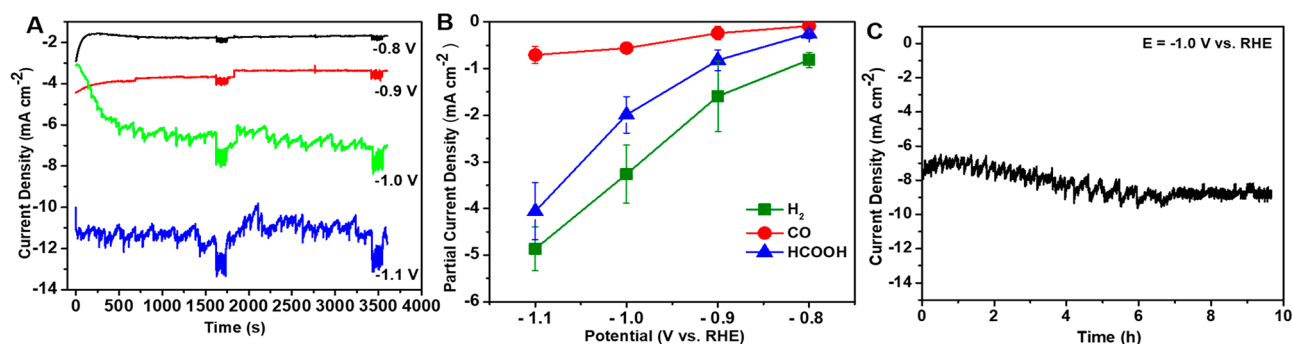


Figure 3. Electrochemical CO₂ reduction performance of the Au₁Sn₂ catalyst: (A) total current density as the function of time at various potentials, (B) HCOOH, CO, and H₂ partial current densities, and (C) long-term stability measured in CO₂-saturated 0.1 M NaHCO₃ at -1.0 V vs RHE. Error bars represent the standard deviation, obtained by studying three different electrodes.

but significant, shift in the Au 4f peaks to higher binding energy was observed with increasing tin content as a result of alloying.³² The bulk composition was analyzed with EDX. The atomic percentages of gold and tin are listed in Table S3, together with the surface composition obtained from XPS. The atomic ratios of constituent metals are close to the stoichiometric molar ratio in the bulk, but not on the surface. It seems that there is always an excess tin on the surface, regardless of the bulk phase composition.

The electrochemical reduction of CO₂ was first studied by linear sweep voltammetry (LSV). The onset potential in the CO₂-saturated solution was notably less negative compared to that in N₂-saturated solution ($E = -0.37$ V in CO₂ and -0.65 V in N₂ vs RHE; see Figure 2A). This observation indicated that an additional process occurred in the CO₂-saturated solution, which requires less overpotential, compared to the one occurring in N₂-saturated solution (i.e., HER).¹² When comparing the voltammetric curves recorded in CO₂-saturated 0.1 mol dm⁻³ NaHCO₃ for the different Au–Sn electrodes, we observed a clear shift in the onset potentials (Figure 2B; also see Figures S9 and S10 for the full LSV curves). Importantly, the trend in the onset potential values does not exactly reflect the change in the composition. This can be rationalized by the fact that new bimetallic phases are formed (rather than simple alloying), which in turn results in a nonlinear change in the bulk and surface energetics, dictating the CO₂ reduction properties. Also note that these differences in the onset

potential are rather substantial as the total range spans through 440 mV!

The CO₂ reduction performance of the bimetallic NPs was evaluated under chronoamperometric conditions. The detected CO₂ reduction products were CO and formate, whereas the remaining charge was attributed to the HER. The total current density values recorded for the Au₁Sn₂ catalyst at different potentials are presented in Figure 3 as an example, together with the partial current density values for the various products. Relatively stable currents were measured during the electrolysis for all compositions, and its value increased with the overpotential (Figure S11). At -1.1 V vs RHE, a stable current of 11 mA cm⁻² was achieved with a FE_{formate} of 42%, whereas the decrease in the current and FE_{formate} was seen at lower overpotentials (with the parallel rise of HER activity). Tuning the composition, however, altered this trend (Figure S11), as a FE_{formate} of 51% was observed at -0.9 V vs RHE over Au₁Sn₄ catalyst, which decreased to 29% at more negative potential (-1.1 V vs RHE). In addition to summarizing the above-mentioned trends for the Au₁Sn₂ catalyst, Figure 3B also depicts that the formation of CO was rather independent from the potential, which translates to a CO/H₂ ratio of 1:6–7. The stability of Au₁Sn₂ catalyst was investigated using a two-compartment cell. The current (8 mA cm⁻²) remained stable within the 10 h window of the experiment (Figure 3C). In fact, a slight increase was witnessed in the current, due to surface roughening of the electrode. TEM analysis confirmed that the catalyst particles retained their morphology, and only a partial

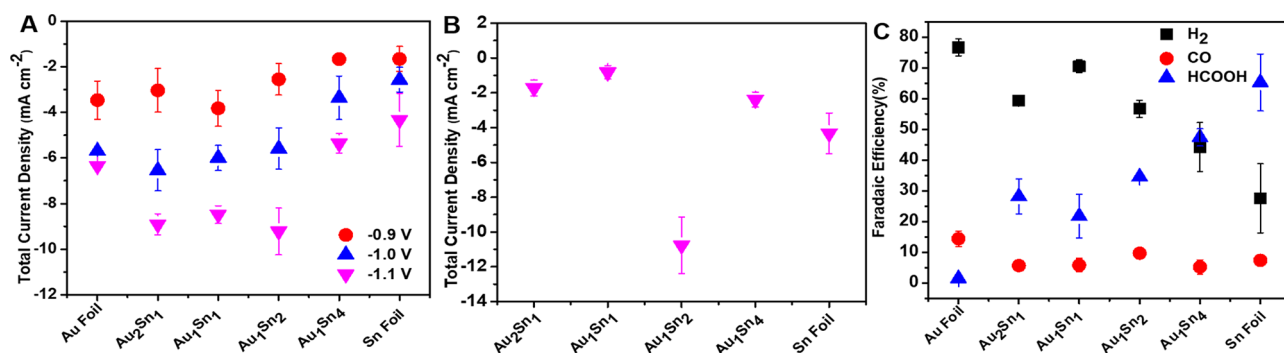


Figure 4. Electrochemical CO₂ reduction activity of Au–Sn NPs, and parent metals. (A) Total current density, (B) normalized total current density measured at –1.1 V vs RHE, and (C) Faradaic efficiency values as a function of composition at –1.0 V vs RHE. Error bars represent the standard deviation obtained by studying three different electrodes.

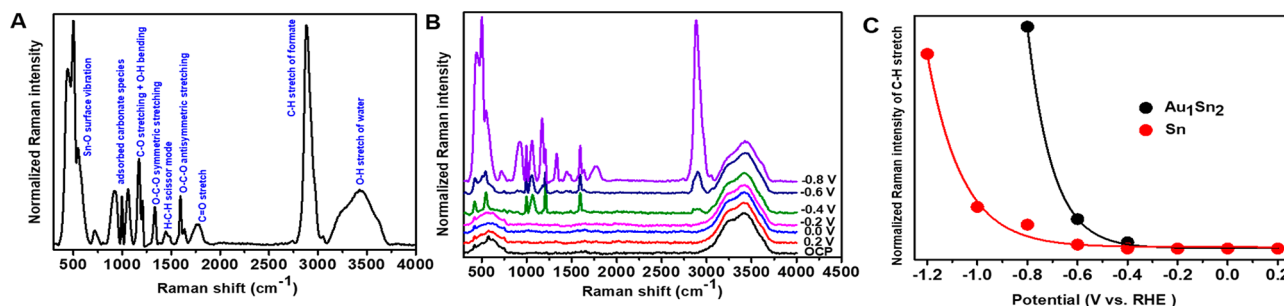


Figure 5. Raman spectra collected on Au₁Sn₂ catalyst in CO₂-saturated 0.1 mol dm⁻³ NaHCO₃: (A) at a potential of –0.8 V vs RHE and (B) as a function of the employed bias potential. (C) Potential dependence for the $\nu(\text{C-H})$ of formate anion at 2880 cm⁻¹ band intensity on Au₁Sn₂ and Sn NP-coated electrodes as a function of the employed bias potential.

reduction of the SnO_x shell was observed from the TEM–EDX analysis (Figure S12), where the initial O/Sn ratio (0.26) dropped to 0.15 after electrolysis (while the Au/Sn ratio remained constant). These data confirm that the crystalline bimetallic core and the amorphous shell are both stable under the reduction conditions, although the latter one undergoes partial reduction.

Figure 4A compares the total current density values recorded for the Au–Sn NPs as well as the parent metals. At less negative potentials (i.e., –0.9 V vs RHE), the gold foil exhibited the largest total current density, whereas at more negative potentials, the bimetallic samples outperformed both gold and tin. The three samples where the AuSn phase was present (see XRD analysis) showed significantly higher total current densities. This trend was further magnified when the current density was normalized with the electrochemically active surface area (Figure 4B); also see the discussion in the Supporting Information. Figure 4C shows product distribution as a function of compositions at –1.0 V vs RHE. Gold foil produces about 80% H₂, whereas CO and formate are produced in relatively small amounts. The Au–Sn NPs produced a considerable amount of formate as the dominate reduction product is CO₂. A relatively linear correlation was observed between the tin concentration in the Au–Sn bimetallic NPs and the formate FE, whereas HER was gradually suppressed to about 44% (adding tin to gold resulted in a slower kinetics toward H₂ production). The FE_{CO}, on average, was about 10%, and there it showed very little composition dependence at this potential. Generally, the H₂/CO ratio varies in a broad range between 2.5 and 10, depending on the composition and the potential (Figure S11). We performed a series of control experiments in which

electrodes were prepared from physical mixtures of Au and Sn NPs. A physically mixed Au + Sn electrode (1:2 molar ratio) with similar loading exhibited a much lower current density and CO₂ reduction selectivity (Figure S13 and discussion therein). Kelvin probe measurements proved that the bimetallic nanoparticles are new chemical entities, having distinctly different electronic properties (Figure S14 and discussion therein). The reducing power (work function) alone cannot explain the trends in the catalytic activity, but there are additional factors to consider.³³

Selective isotopic labeling experiments were performed to gain insights into the mechanism of CO₂ reduction on Au–Sn catalysts (see also Figure S15 and discussion therein). We found that the produced CO and formate originate from the aqueous CO₂ supplied primarily through fast equilibrium with the bicarbonate ions in the close vicinity of the electrode rather than the purged CO₂. Although the obtained trends are similar to those observed for CO₂ reduction on Au-, Cu-, and N-doped carbon surfaces,^{34–36} here, we confirmed a similar pattern for the production of formic acid. To gain further insights on the mechanism of the CO₂ reduction process, Raman spectra were collected under electrochemical control. This allows the direct observation of reaction intermediates and/or products as they are produced during the electrochemical reaction.^{37,38} The spectra collected between the open circuit potential and –0.2 V exhibit only bands belonging to SnO_x (482, ~623, ~772 cm⁻¹)³⁹ and the O–H stretching mode of the adsorbed water (3000–3700 cm⁻¹, note that this band was almost independent from the potential).⁴⁰ At –0.4 V, new bands started to appear and their intensities gradually increased at more negative potentials. The SnO_x bands became more intense and slightly shifted due to surface defects as a

result of partial reduction.^{41,42} At potentials more negative than -0.8 V, it was quite difficult to collect Raman spectra because of the intense gas evolution. The assignments of all bands are shown in Figure 5A and summarized in Table S4. At -0.8 V, the spectrum showed a strong band at 2880 cm^{-1} and several bands of medium and weak intensity in the region of $900\text{--}1774\text{ cm}^{-1}$. These bands are similar to those observed during adsorption of formic acid on silver colloids and Cu,^{38,43} indicating the presence of formate-related and adsorbed bicarbonate species. Although the spectra recorded for Au₁Sn₂ and Sn NPs were similar (Figures 5B and S16C), higher overpotential was required for developing the bands on Sn NPs (Figure 5C), consistent with the observed shift in the onset potential on the LSV profiles (Figure 2B).

In summary, this study proved the advantage of intermetallic phases compared to Au and Sn NPs as well as their physical mixtures. The Au₁Sn₂ catalyst (containing almost pure AuSn phase) showed the lowest overpotential for CO₂ reduction, 400 mV less negative compared to Sn! Under optimal conditions, formate with high efficiency in the liquid phase and simultaneous syngas in the gas phase were obtained, as two high-value products with reasonable current density (up to 10 mA cm^{-2}). Comparing the activity descriptors (i.e., overpotential, current density, and product distribution) with those of other bimetallic nanoparticles in Table S1, we can conclude AuSn catalysts are indeed very promising. The superior catalytic behavior is related to the changes in the adsorption site, surface energy, and orientation of the adsorbed species. Selective isotopic labeling experiments were performed under non-equilibrium conditions, suggesting that CO₂ supplied through fast equilibrium with the bicarbonate, rather than CO₂ in the bulk solution, is the primary source of the produced CO and formate. Raman spectroelectrochemistry proved the presence of bicarbonate anions on the electrode surface under reaction conditions and confirmed the generation of formate anions at notably less negative potential on the AuSn phase compared to the pure Sn electrode.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: [10.1021/acsenergylett.8b01996](https://doi.org/10.1021/acsenergylett.8b01996).

Experimental methods, TEM and SEM images, XRD analysis, and additional electrochemical measurements (PDF)

■ AUTHOR INFORMATION

Corresponding Author

*E-mail: janaky@chem.u-szeged.hu. Twitter: @JanakyLab.

ORCID

Csaba Janáky: 0000-0001-5965-5173

Notes

The authors declare no competing financial interest.

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